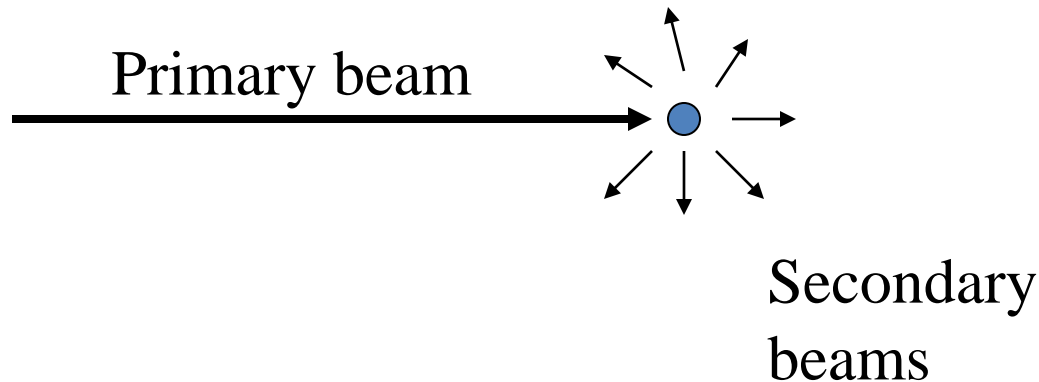


# Part 1

## X-ray Crystallography

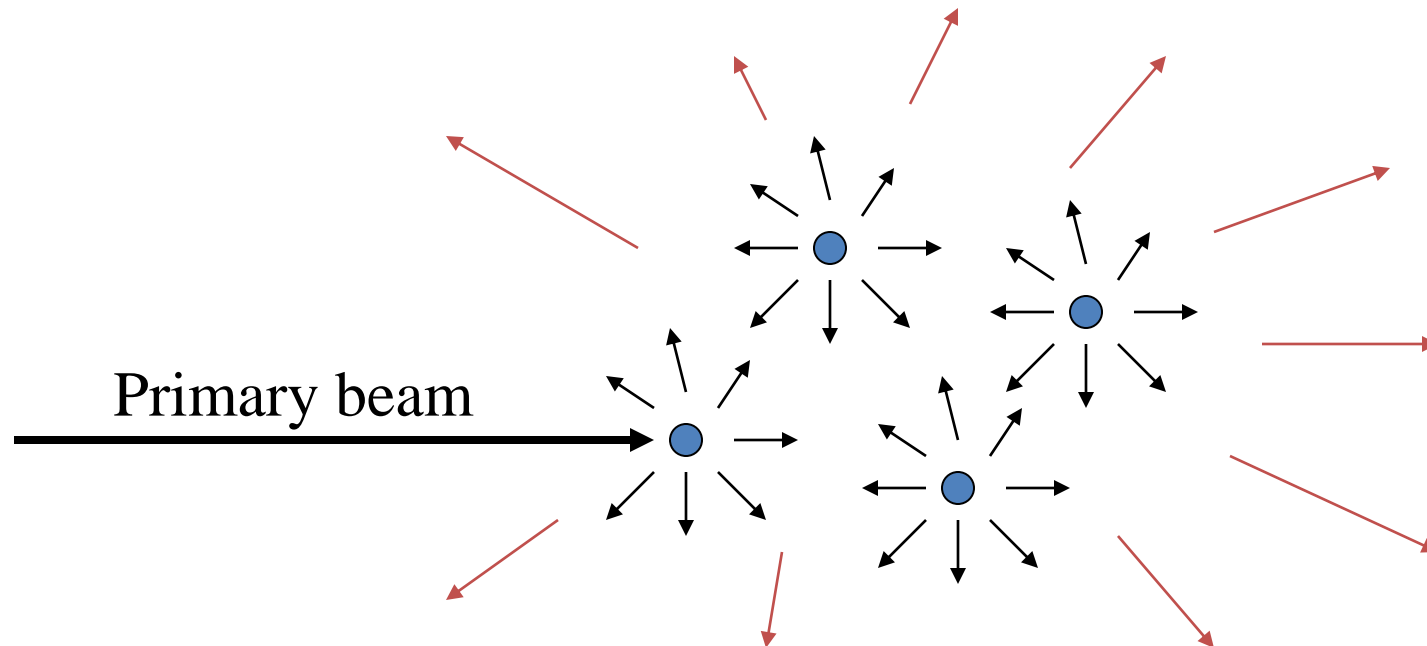
# What happens to electron when it is hit by x-rays?

1. The electron starts vibrating with the same frequency as the x-ray beam
2. As a result, secondary beams will be scattered in all directions



# Scattering from a molecule

1. Molecule is composed of many electrons
2. Each electron will scatter secondary radiation upon exposure to x-rays
3. The scattered secondary beams will interact and cause interference
4. The scattering from a molecule is dependent upon the number of and distances between electrons
5. In other words, scattering from molecule is dependent on its atomic structure
6. If we would know the amplitudes and phases of scattered molecule, we could calculate the structure of molecule...

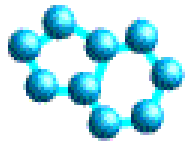


# In practice...

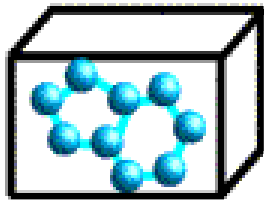
- Scattering from a single molecule is far too weak to be observed but possible.
- If molecules are all oriented in the same way (like in crystal), the scattering from individual molecules will be multiplied in certain directions

# The unit cell and a crystal

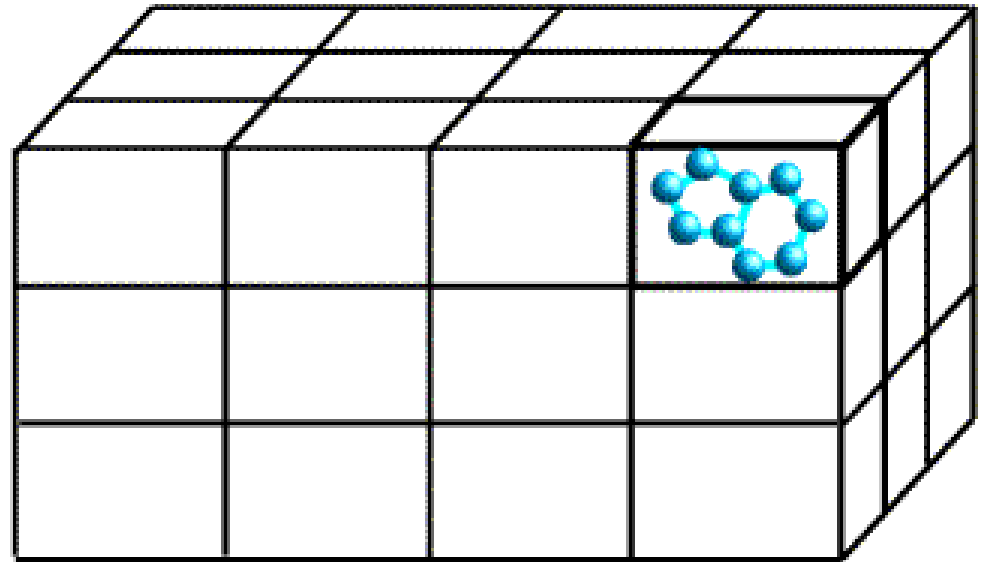
molecule

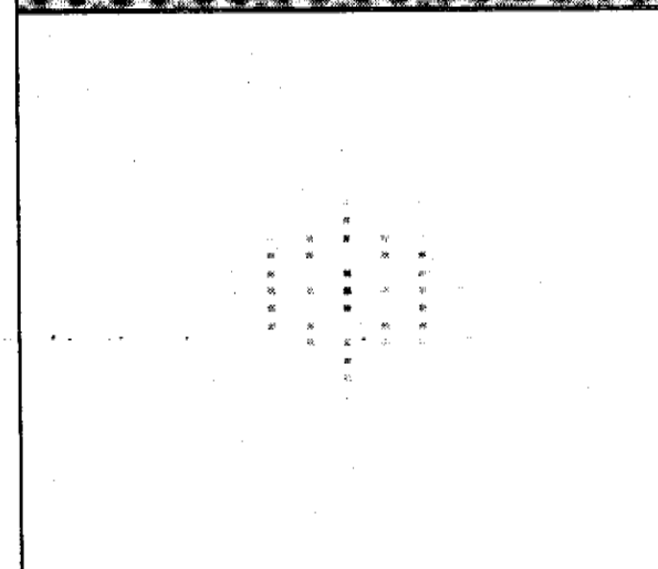
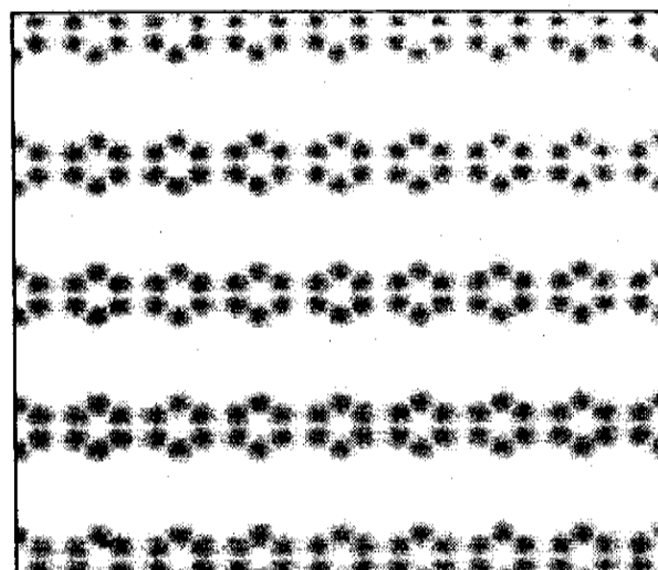
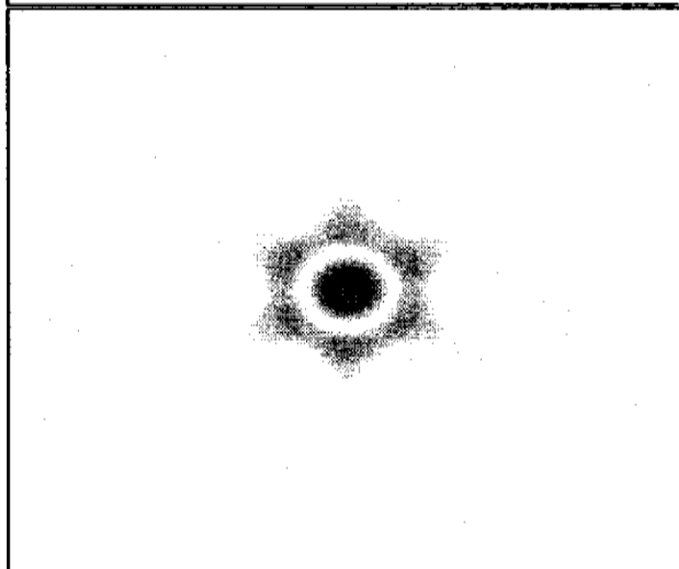
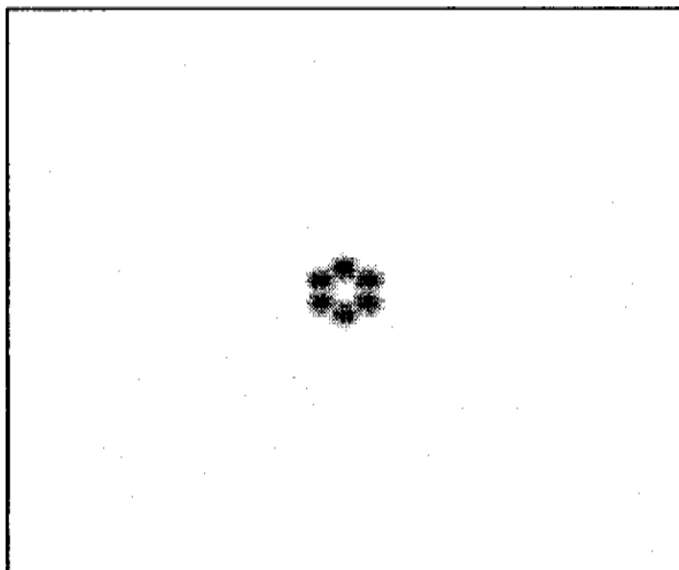


unit cell



crystal

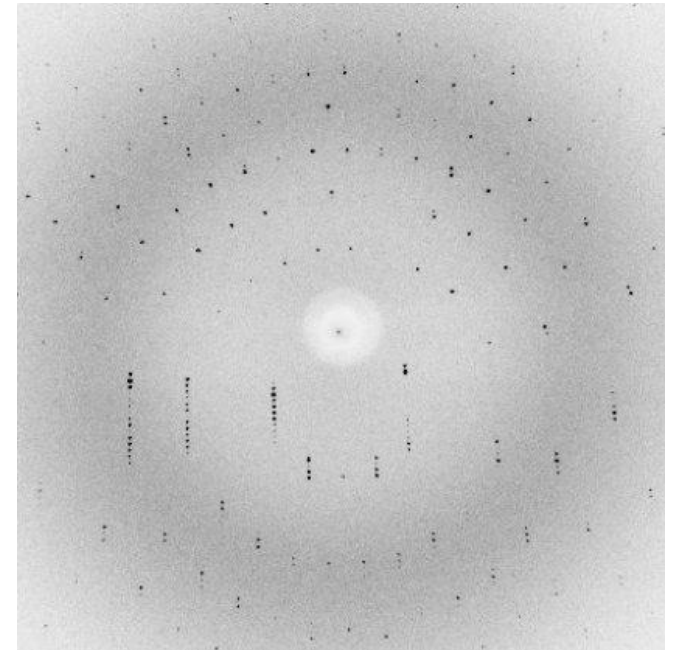


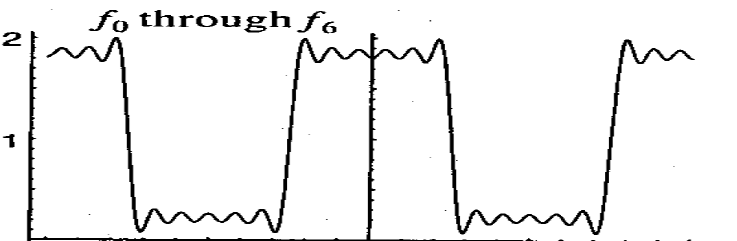
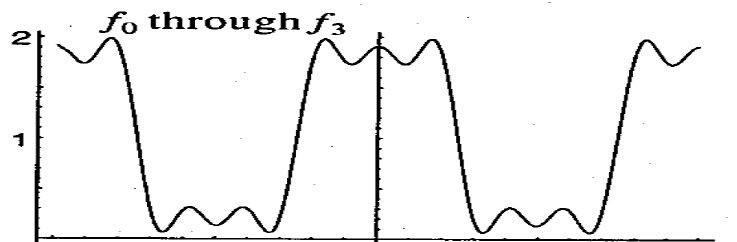
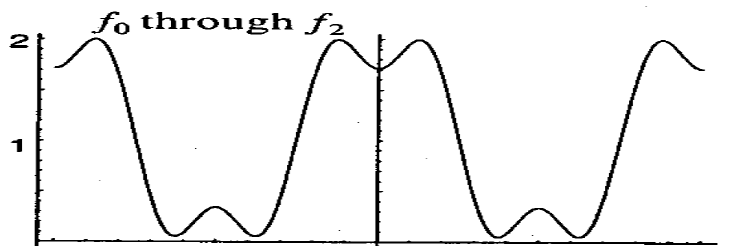
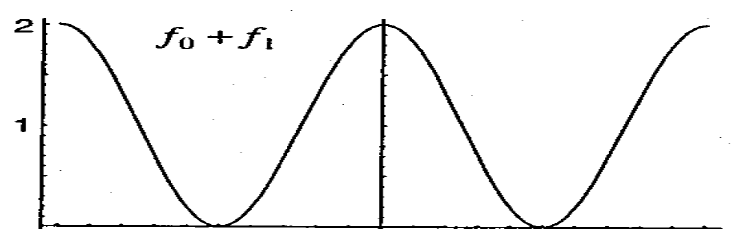
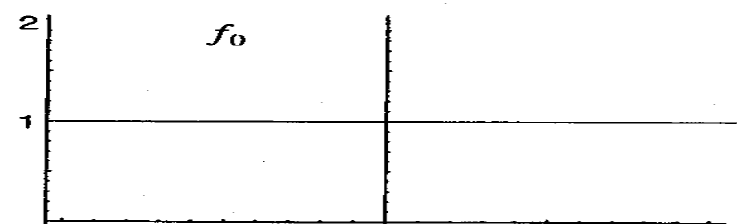
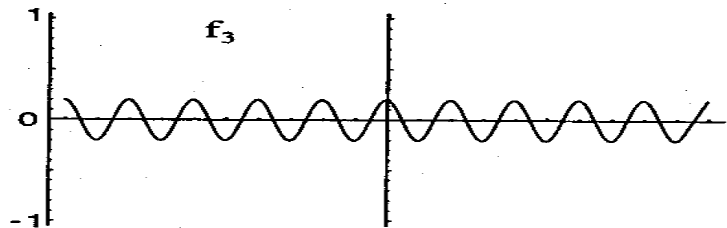
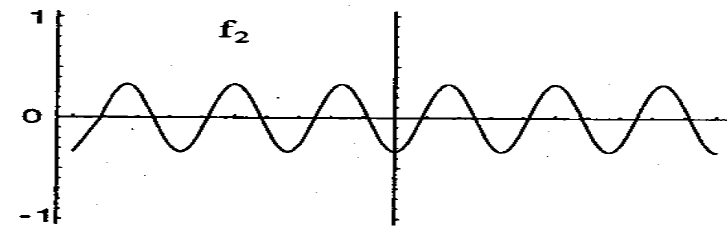
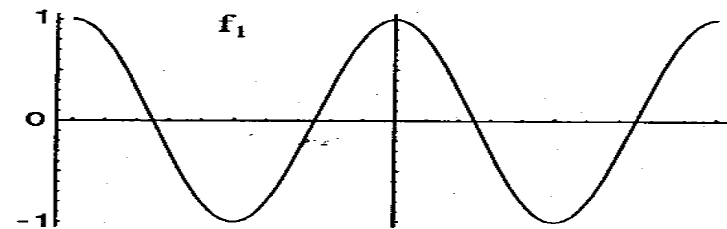
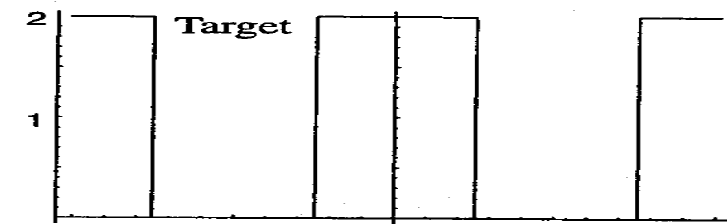


# The electron density equation

$$\rho(xyz) = \frac{1}{V} \sum_h \sum_k \sum_l F(hkl) \exp[-2\pi i(hx + ky + lz) + i\alpha(hkl)]$$

- $h, k, l$  – indices of reflections
- $xyz$  – coordinates
- $F$  – amplitude of reflections
- $\alpha$  – phase of reflections
- $V$  – unit cell volume

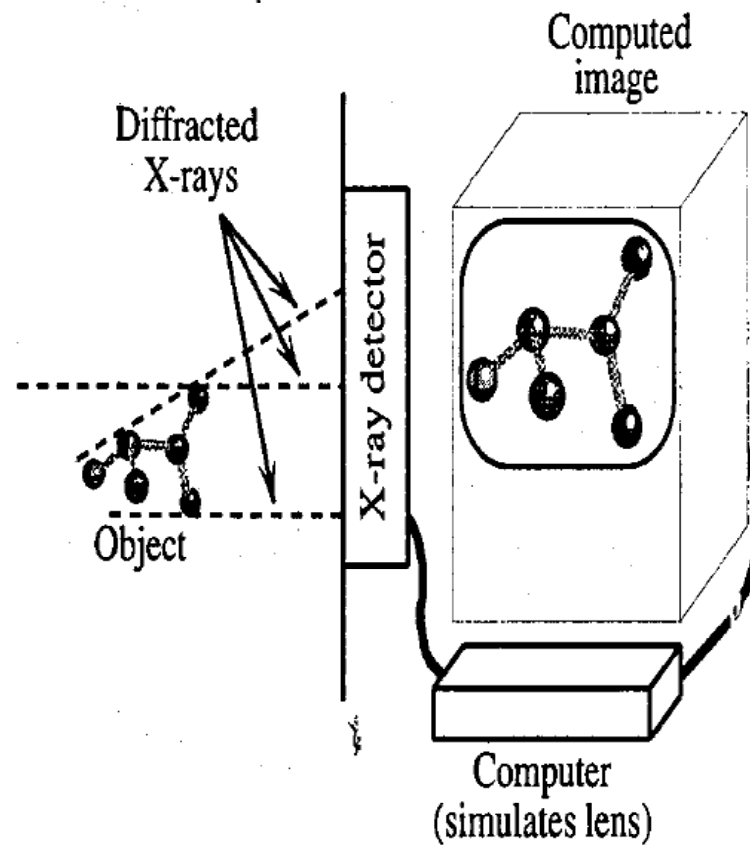
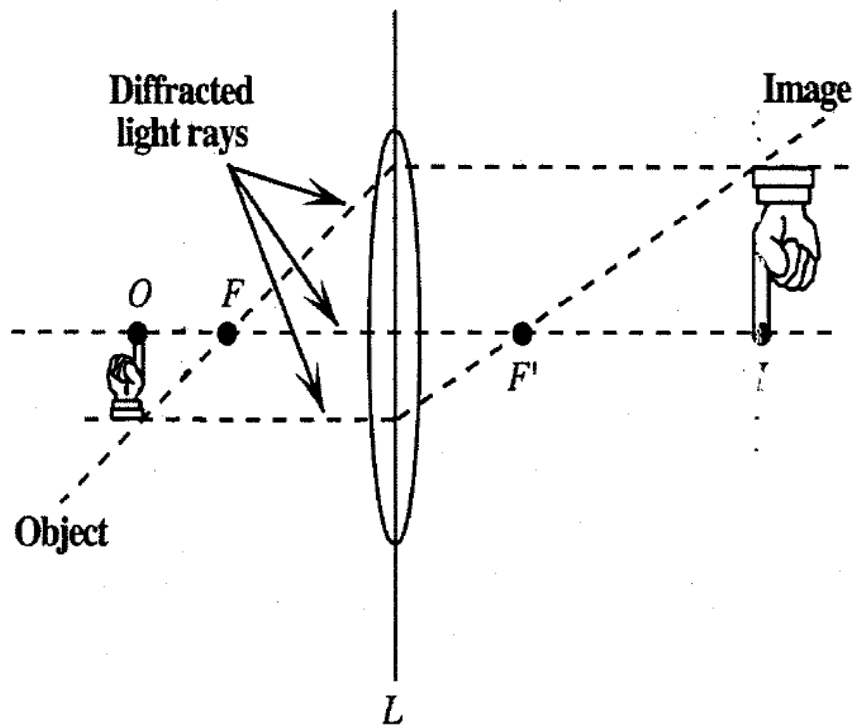






# The Phase Problem

- With detector you can measure only the intensity of reflections
- The information about phases is lost – there is no such thing as “phase meter” because we have no lenses for x-rays.
- This means, you must obtain phase information in some other way
- For small molecules (<100 atoms), direct methods exist. This means, that you can calculate phases from amplitudes without any extra information.
- Proteins are far too big to use direct methods, so other tools are developed



# Isomorphous replacement

- By introducing heavy atoms in protein crystal (by soaking), the diffraction pattern can be altered
- It is possible to determine positions of heavy atoms and from them the phases
- One must use at least 2 different heavy atom soaks
- Problems:
  - 1) Unit cell dimensions of crystal might change upon soaking
  - 2) Crystal might get destroyed upon soaking and not diffract at all
  - 3) Heavy atom ions might not bind in well defined places

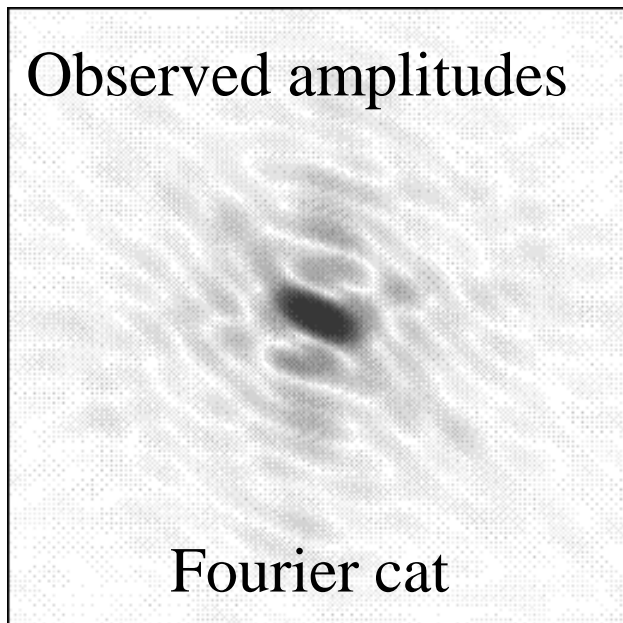
# Molecular replacement

- Currently the most common technique
- Applicable only if a similar structure already exists (at least 25% sequence identity)
- The phases of known structure are combined with intensities of unknown
- Before that, the known model has to be *in silico* placed in an artificial unit cell in the same orientation and translation from origin as in the structure of interest
- For this, rotation and translation functions exist

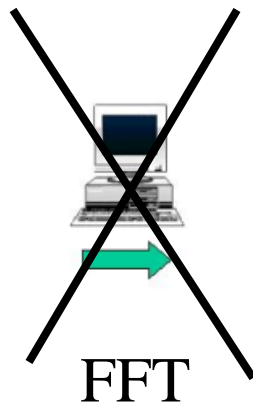
## Problems:

- May not work, if unknown structure is less than 30 % identical to the known structure
- Model bias – what's that?

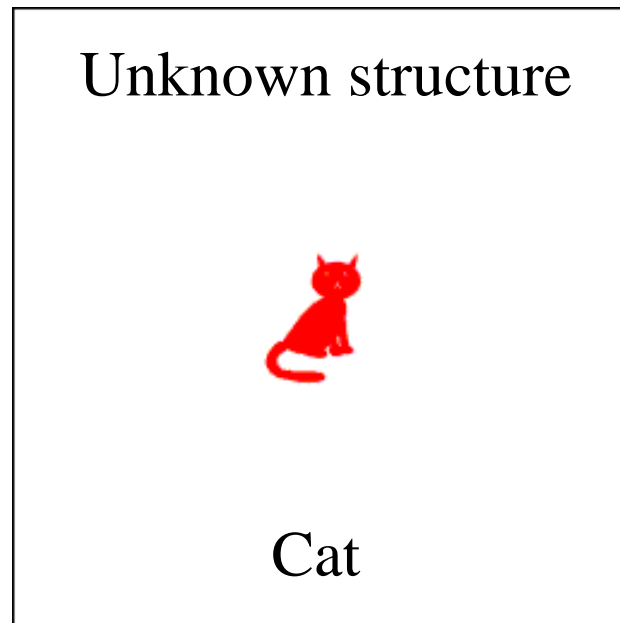
Observed amplitudes



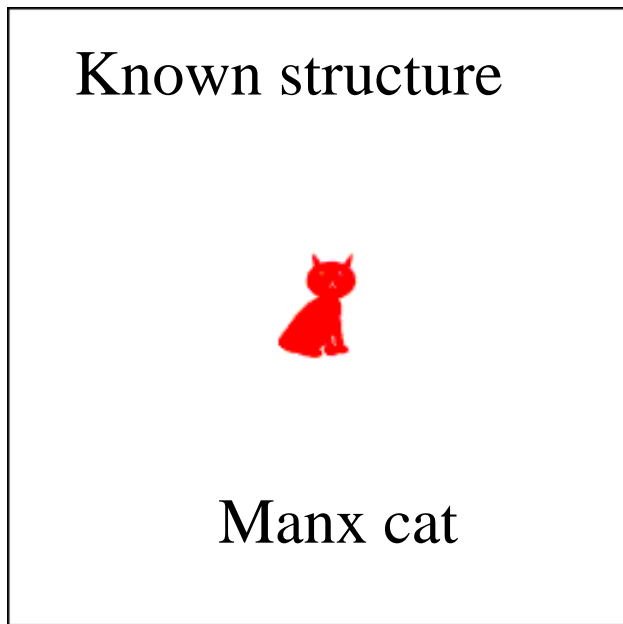
Phases  
unknown!



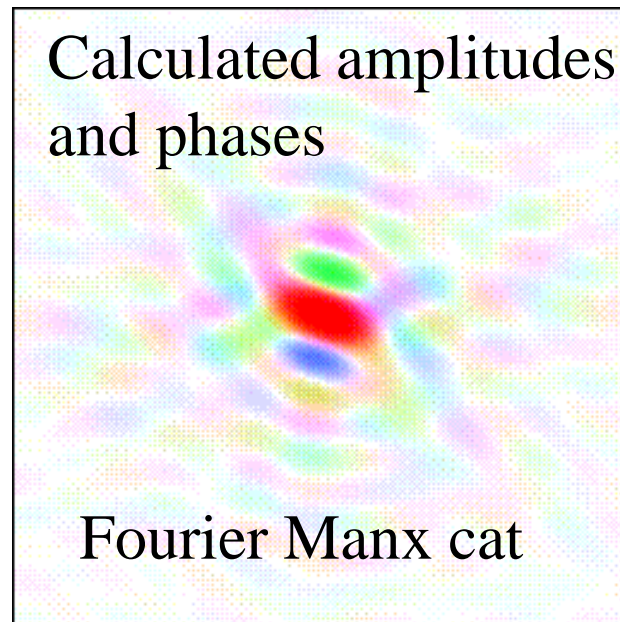
Unknown structure



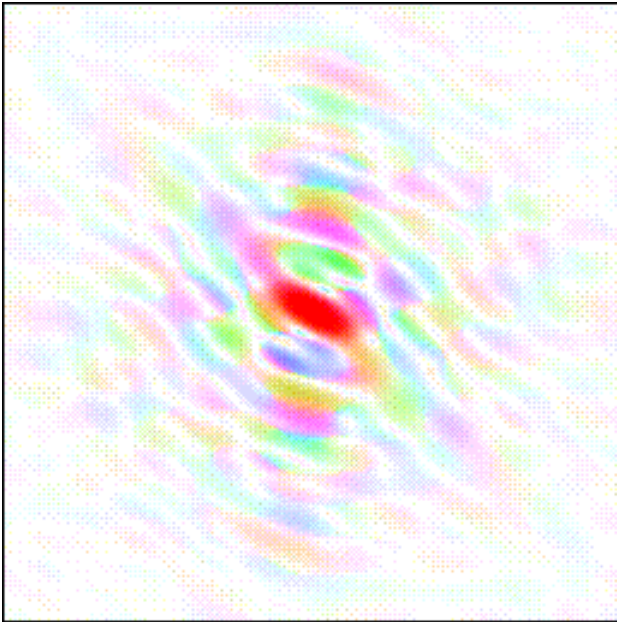
Known structure



Calculated amplitudes  
and phases

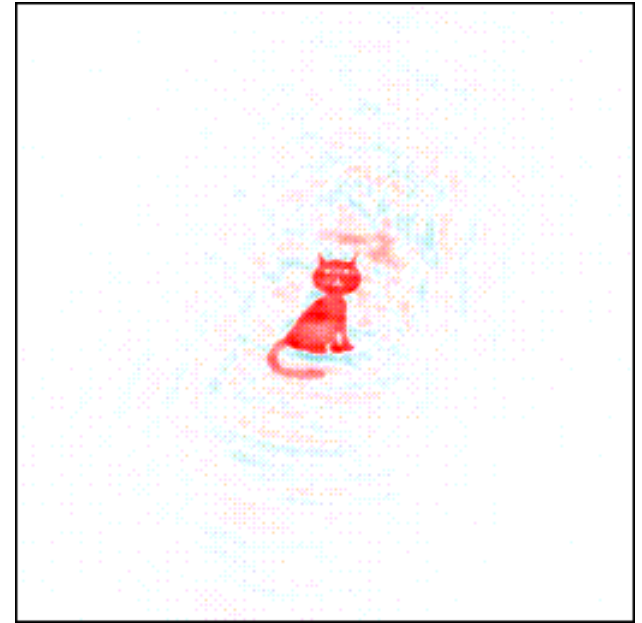


Observed amplitudes,  
calculated phases



FFT

The tail becomes visible!



Be aware – this happens, if structures  
are not similar enough!!



Duck



Fourier duck

Duck amplitudes + cat  
phases

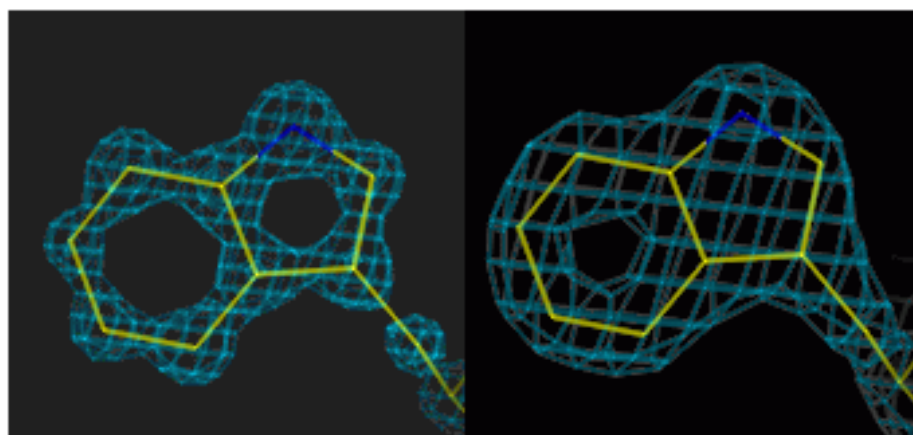


Looks like a cat!!

# Model building

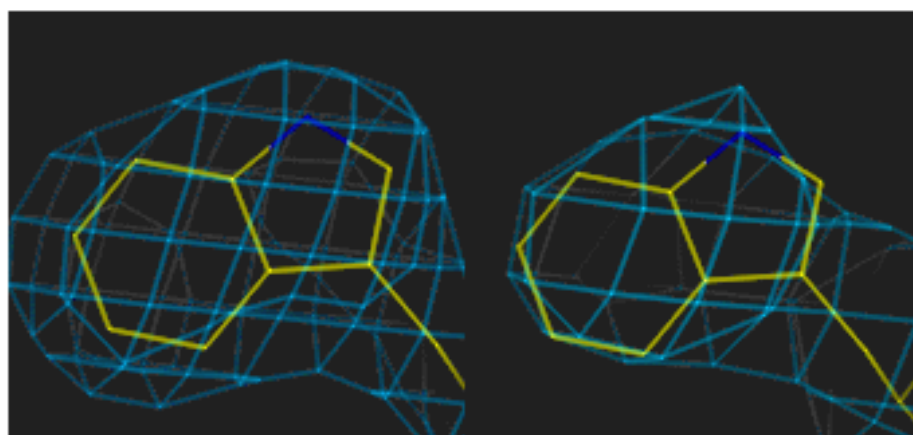
- Fitting of protein sequence in the electron density
- Easy in molecular replacement
- More difficult if no initial model is available
- Unambiguous if resolution is high enough (better than 3.0 Å)
- Can be automated, if resolution is close to 2Å or better





1.0Å

2.5Å

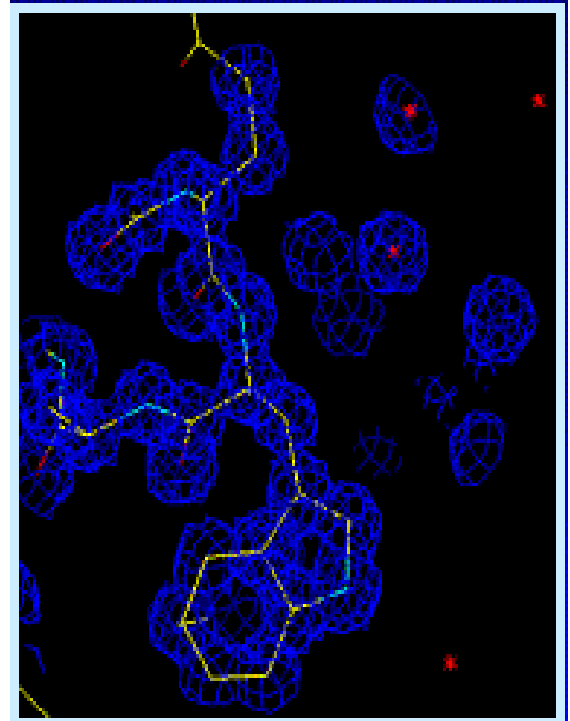
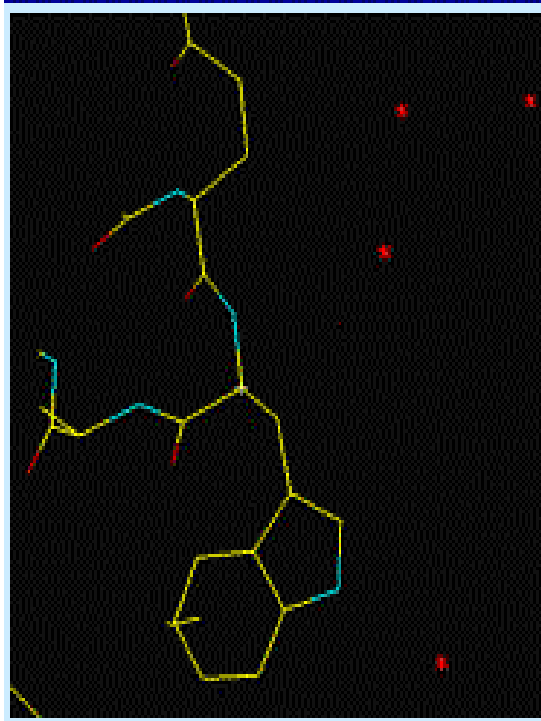
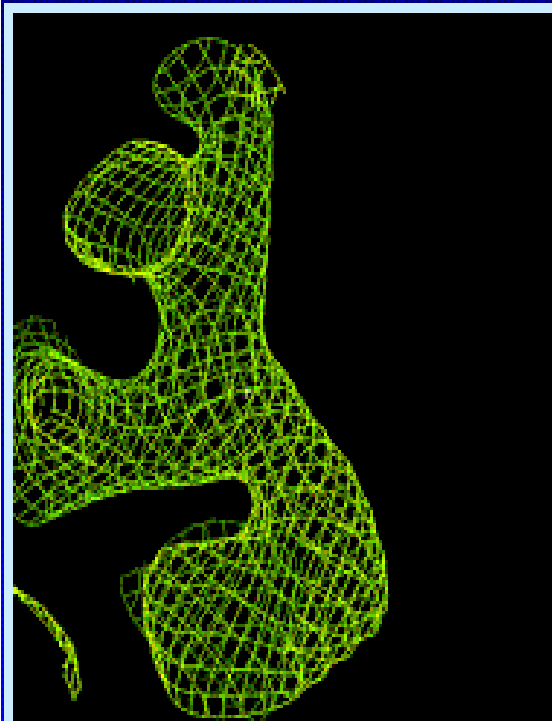


3.0Å

4.0Å

# Refinement

- Automated improvement of the model, so it explains the observed data better
- The phases get improved as well, so the electron density maps get better



# Validation

- Assessment of the final(?) model quality
- How the geometry of amino acids look like? (Ramachandran plot)
- Are non-covalently atoms far enough from each other? (no atom bumps)
- Are residues “happy” in their environment? (hydrophobic in core, polar on surface)
- Are the hydrogen donors/acceptors satisfied?

# Part 2

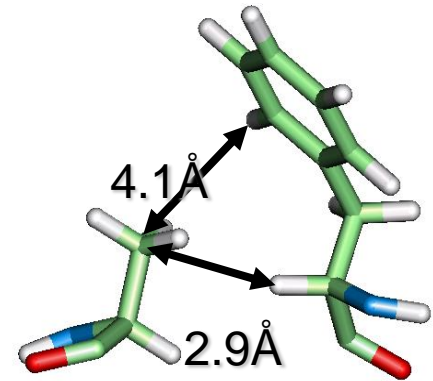
## NMR Structure Determination

# NMR Interactions

## NOE

- a through space correlation ( $<5\text{\AA}$ )
- distance constraint

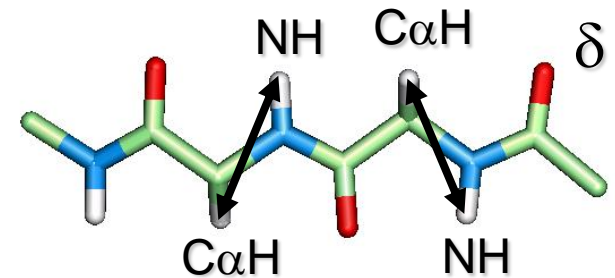
NOE



## Coupling Constant (J)

- through bond correlation
- dihedral angle constraint

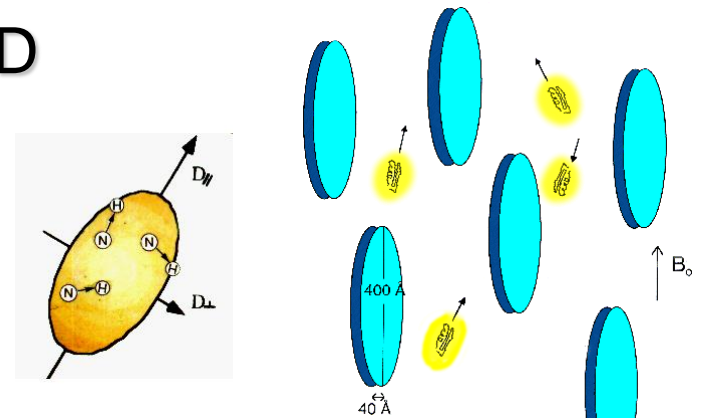
J



## Chemical Shift ( $\delta$ )

- very sensitive to local changes in environment
- dihedral angle constraint

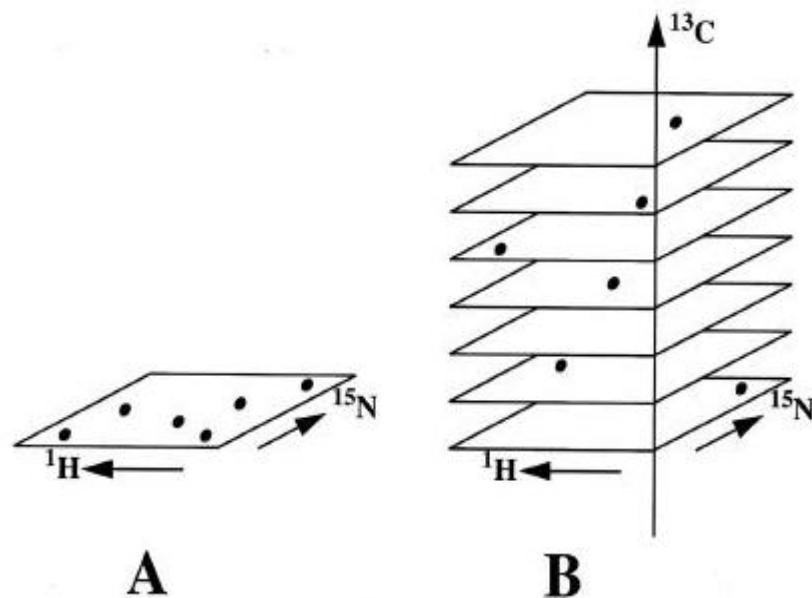
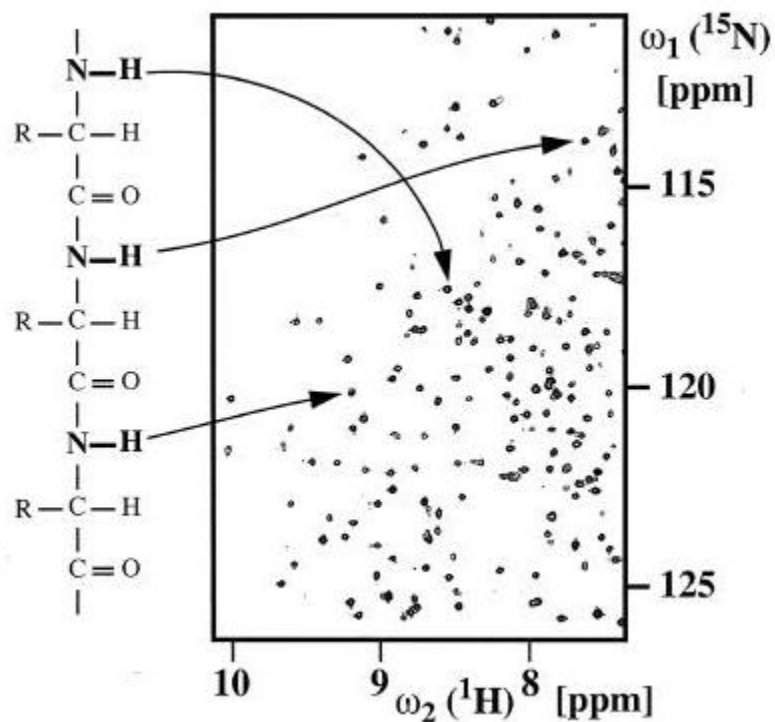
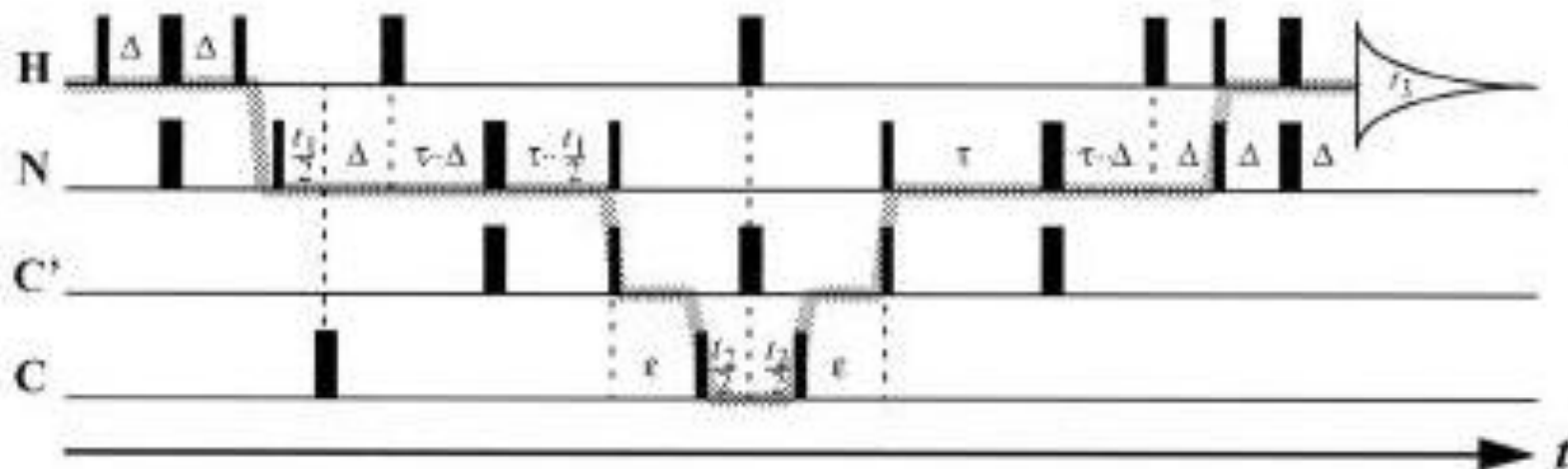
D



## Dipolar coupling constants (D)

- bond vector orientation relative to magnetic field
- alignment with bicelles or viruses

# COSY- COrrrelation SpectroscopY



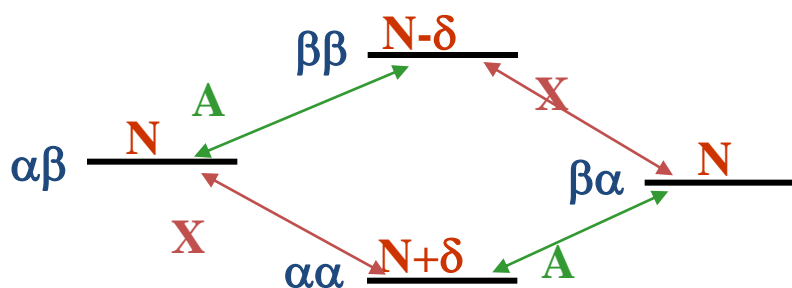
# NOE- Nuclear Overhauser Effect

**Nuclear Overhauser Effect (NOE, h)** – the change in intensity of an NMR resonance when the transition of another is perturbed, usually by saturation.

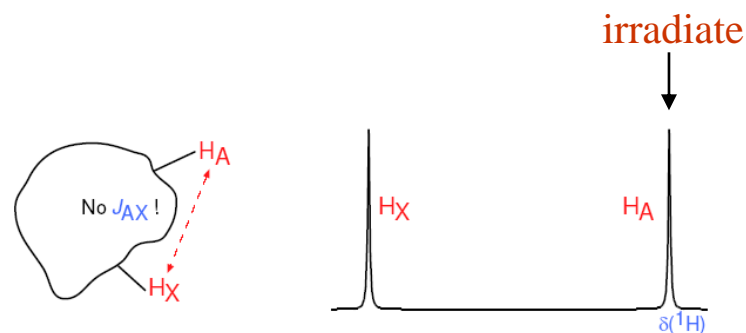
$$\eta_i = (I - I_o)/I_o$$

where  $I_o$  is thermal equilibrium intensity

**Saturation** – elimination of a population difference between transitions (irradiating one transition with a weak RF field)

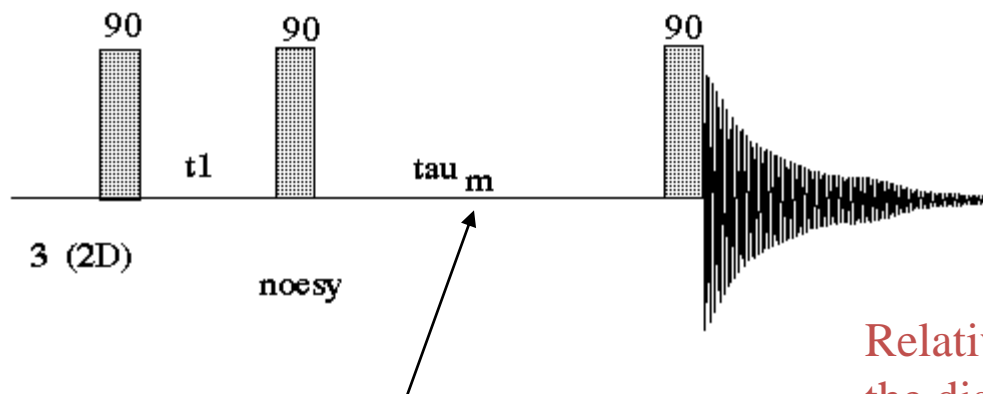


Populations and energy levels of a homonuclear AX system (large chemical shift difference)



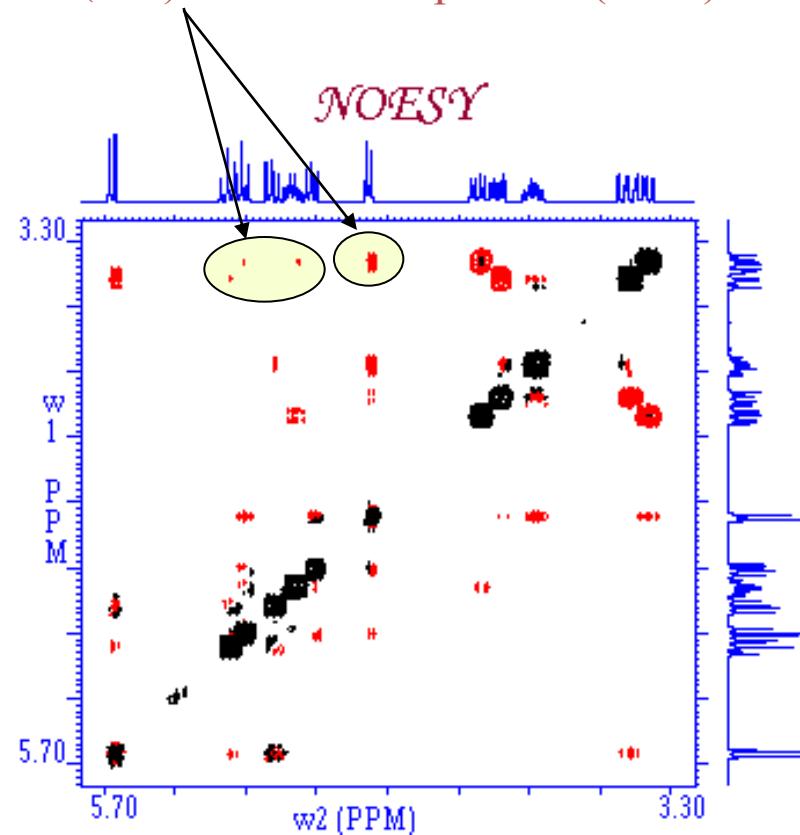
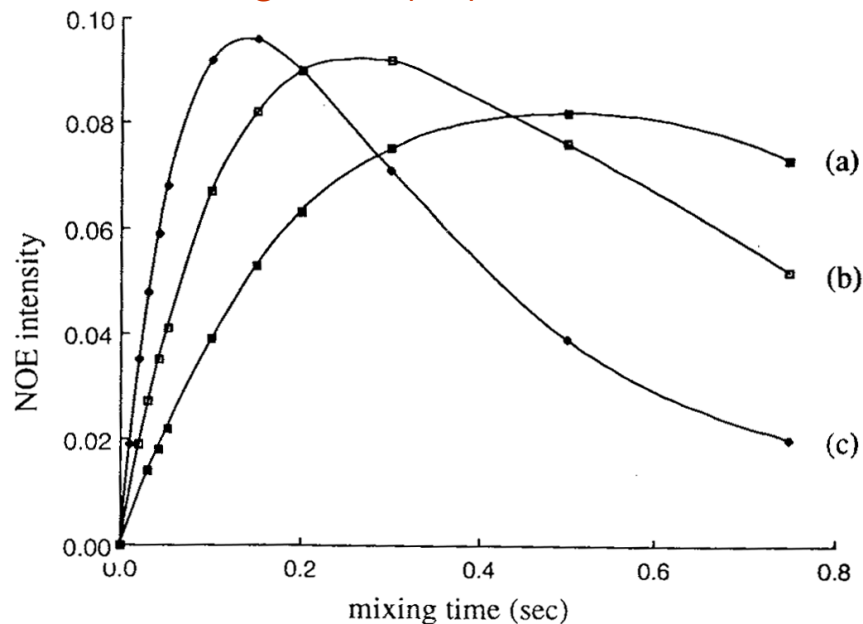
Observed signals only occur from single-quantum transitions

# 2D NOESY (Nuclear Overhauser Effect)



Relative magnitude of the cross-peak is related to the distance ( $1/r^6$ ) between the protons ( $\geq 5\text{\AA}$ ).

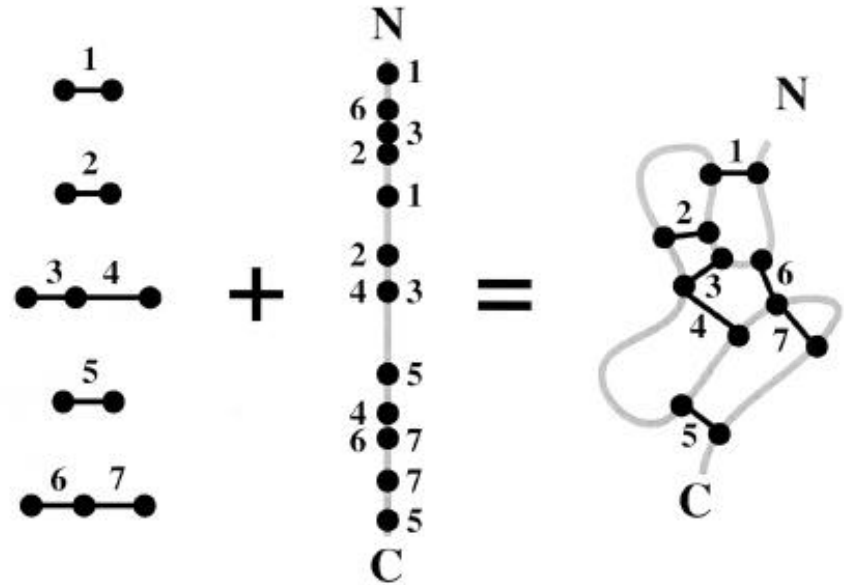
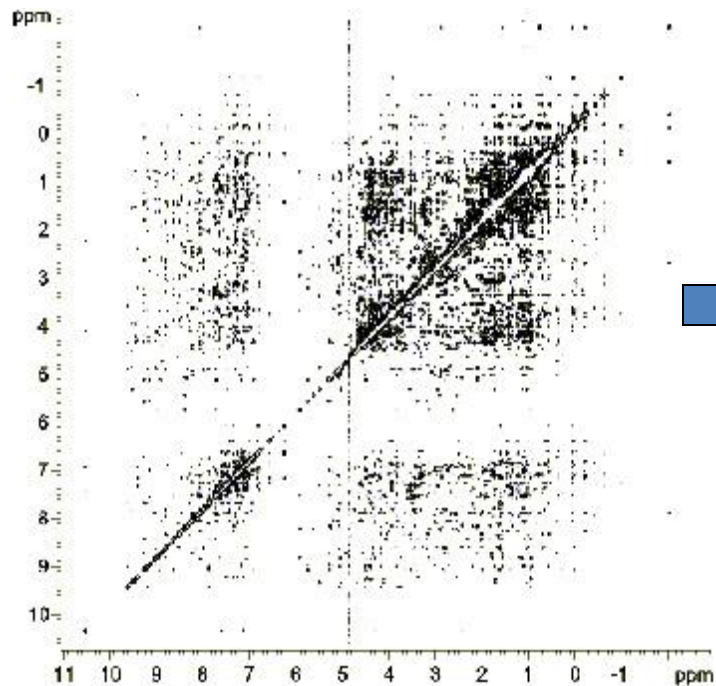
NOE is a relaxation factor that builds-up during  
The "mixing-time ( $t_m$ )"





# NMR Structure Determination

*Going from NOESY Data to a structure*

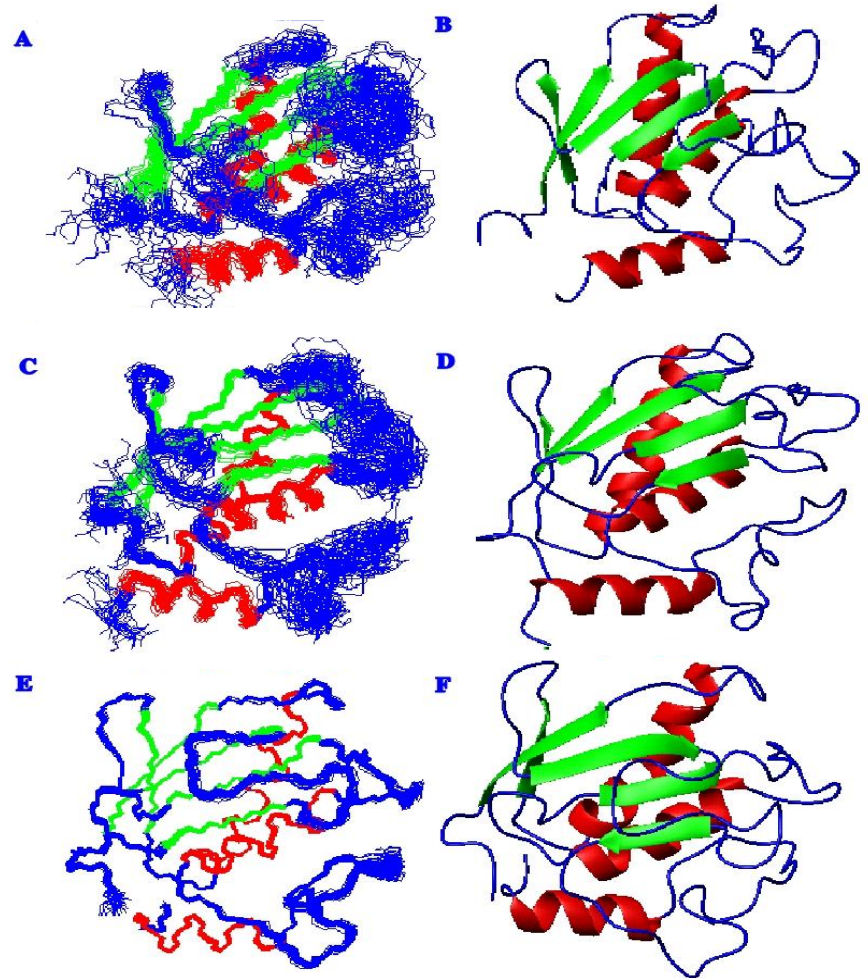
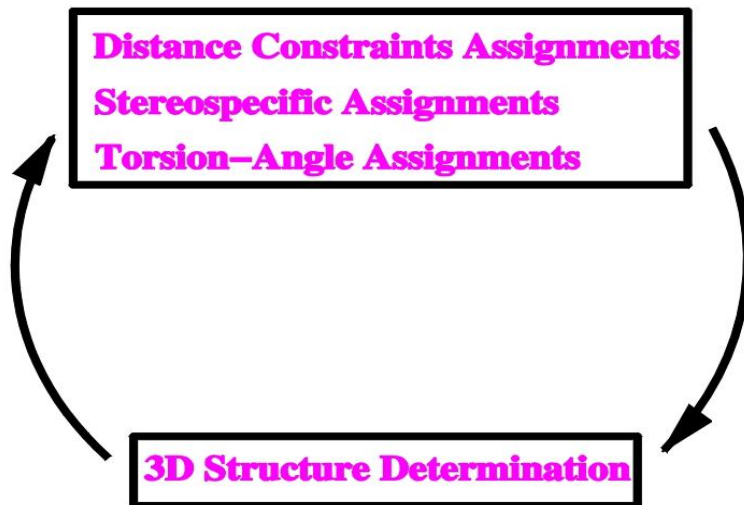


2D NOESY Spectra at 900 MHz

*Once we complete the Assignment of All the Remaining NOEs in an Iterative Process to Obtain the Structure*

## **Protein Structure Refinement**

### **Iterative Cycle**

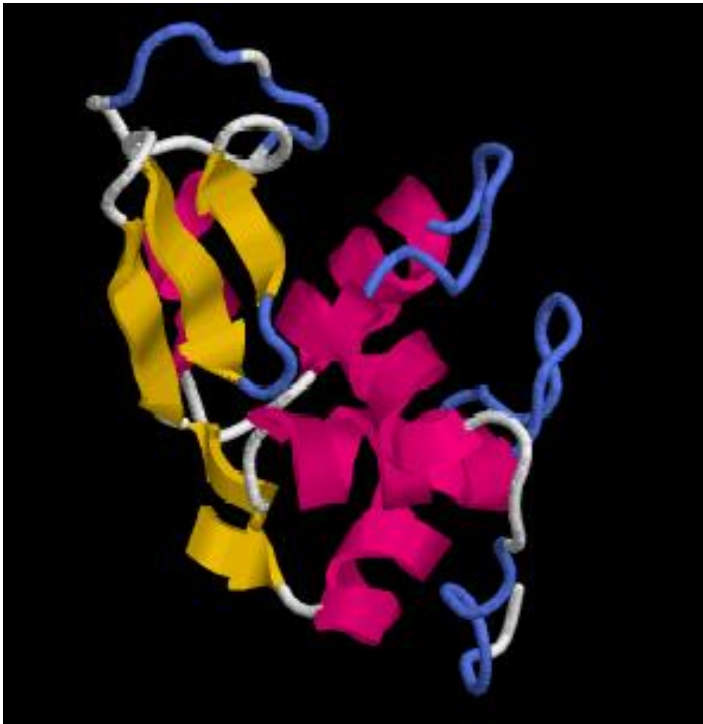


**More Constraints the Better the Structure**

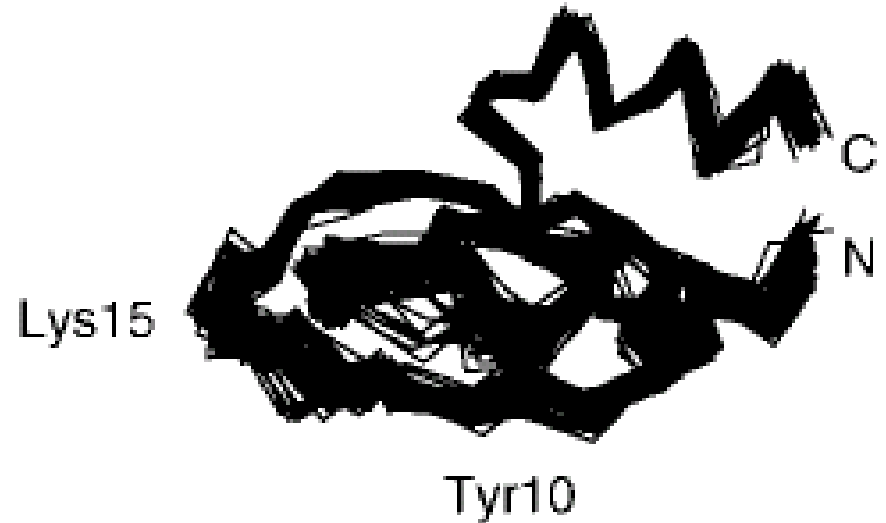
# NMR Structure Determination

## *Two Very Important Facts to Remember*

- NOEs Reflect the Average Distance
- Protein Structures Are Dynamic



We visualize protein structures as a static image

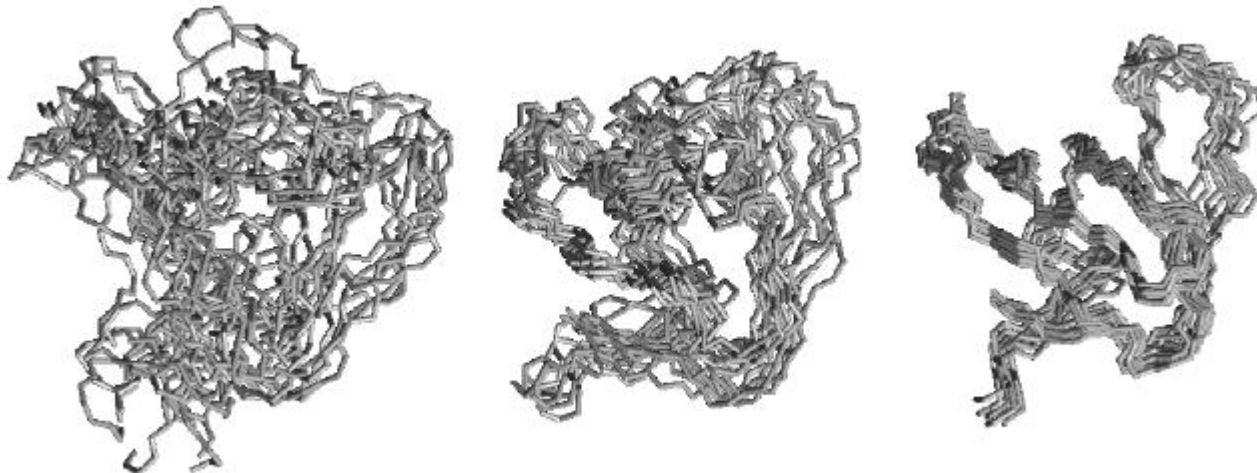


In reality, protein undergoes wide-ranges of motions (snapshots of 100 BPTI conformations)

# Quality of NMR structures

## *Improving the Quality of NMR Structures*

- Stereospecific Assignments
  - Making stereospecific assignments increase the relative number of distance constraints while also tightening the upper bounds of the constraints
  - There is a direct correlation between the quality of the NMR structure and the number of distance constraints
    - ◆ more constraints → higher the precision of the structure



*Increasing Number of NOE Based Constraints*



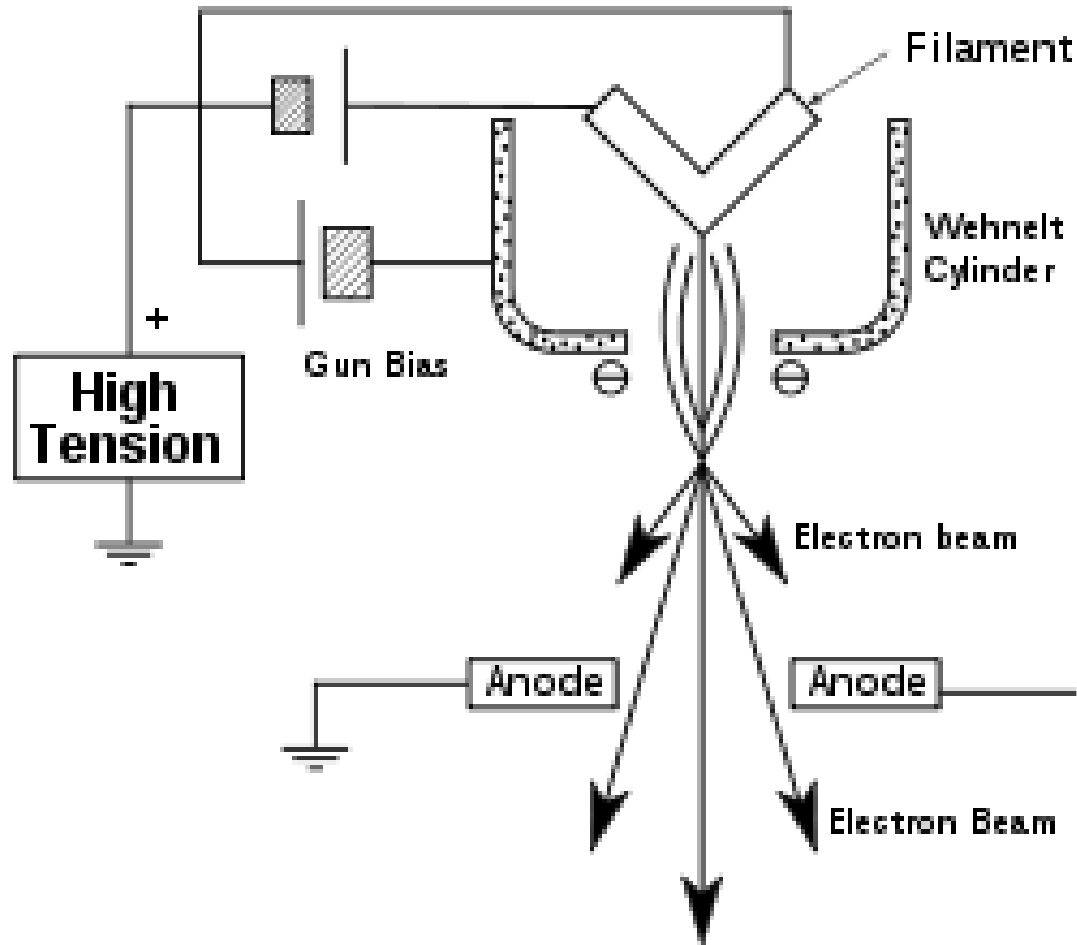
# Part 3

## Structure Determination by Electron Microscopy

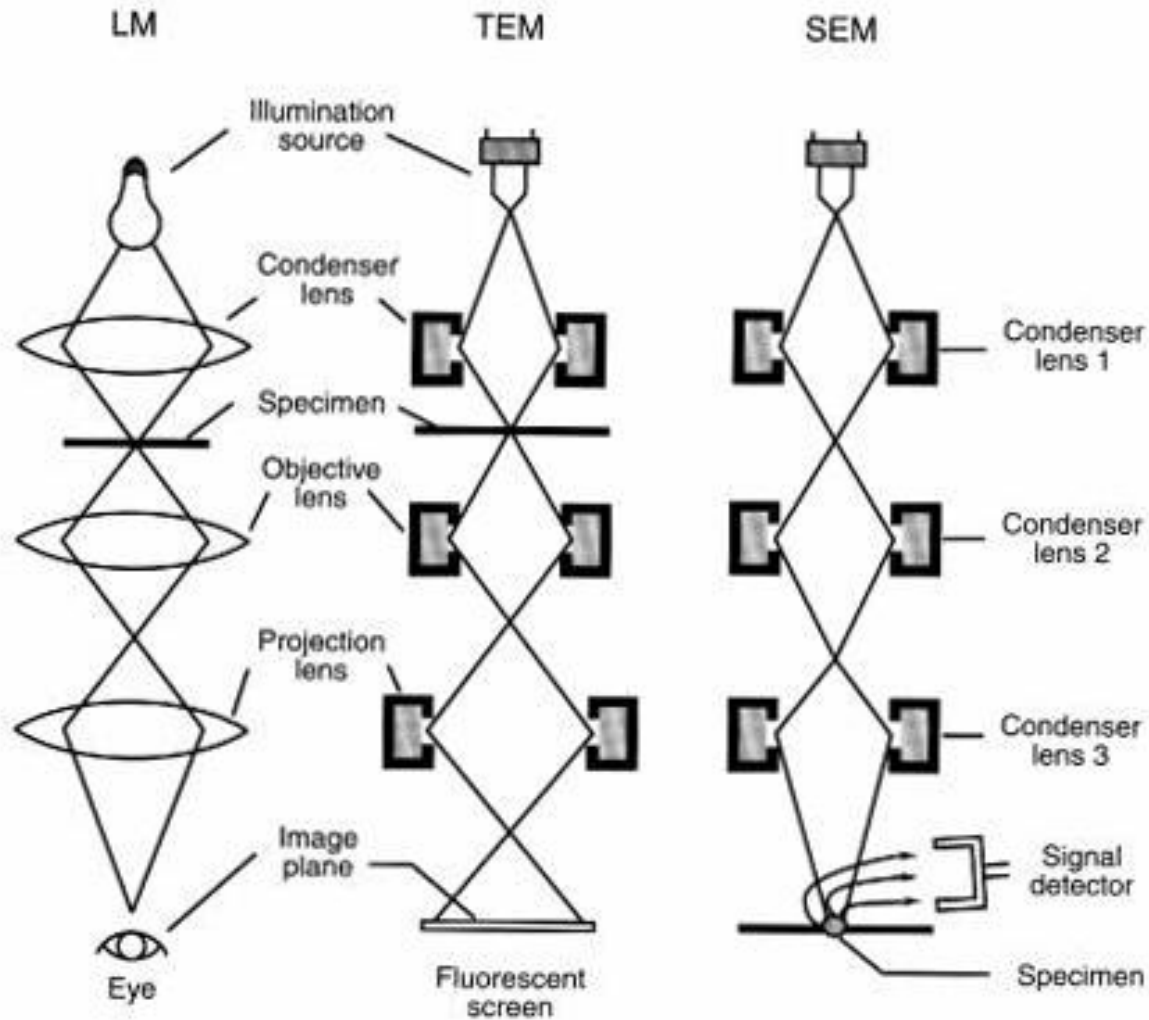
# References and other useful material

- Texts
  - Biophysical Electron Microscopy: Basic Concepts and Modern Techniques by [U. Valdre](#) (Editor), [Peter W. Hawkes](#) (Editor)
  - Three-Dimensional Electron Microscopy of Macromolecular Assemblies by [Joachim Frank](#)
  - Negative Staining and Cryoelectron Microscopy: The Thin Film Techniques by [Robin J. Harris](#), [James R. Harris](#)
- Reviews
  - Henderson, R. The potential and limitations of neutrons, electrons and X-rays for atomic resolution microscopy of unstained biological molecules. *Q Rev Biophys* **28**, 171-93 (1995).
  - Glaeser, R. M. Review: electron crystallography: present excitement, a nod to the past, anticipating the future. *J Struct Biol* **128**, 3-14 (1999).
  - Stowell, M. H., Miyazawa, A. & Unwin, N. Macromolecular structure determination by electron microscopy: new advances and recent results. *Curr Opin Struct Biol* **8**, 595-600 (1998).
- Web
  - <http://ncmi.bcm.tmc.edu/%7Estevel/spintro/siframes.htm>
  - [http://cryoem.berkeley.edu/~nieder/em\\_for\\_dummies/](http://cryoem.berkeley.edu/~nieder/em_for_dummies/)

# Thermionic emission (Shottky effect)



# Electron lenses and “optical path”





# Why use electrons....Part 1



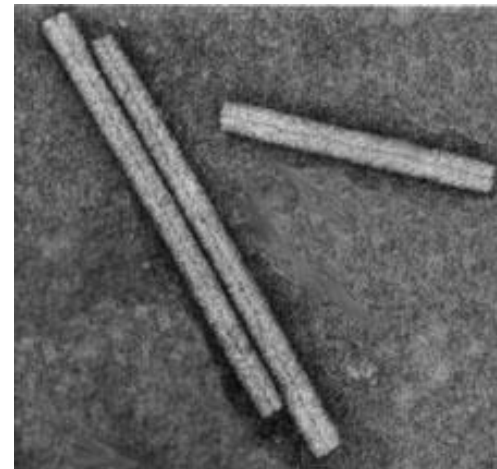
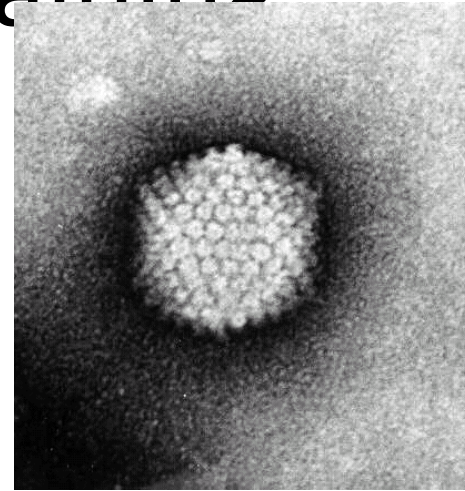
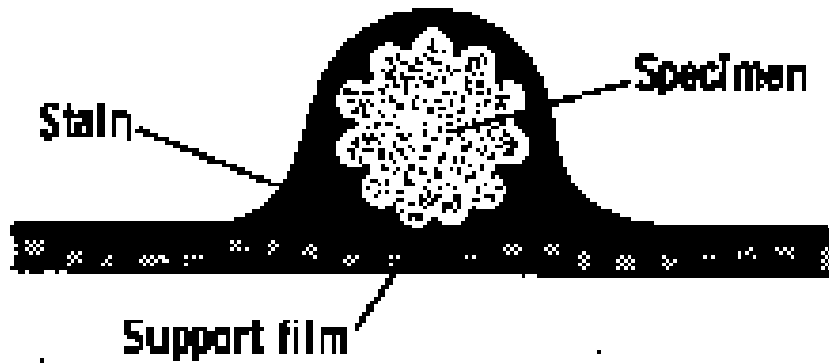
# Why use electrons...Part 2

	Electrons 80-500keV	X-rays 1.5 A
Ratio inelastic/elastic	3	10
Mechanism of damage	2 <sup>nd</sup> e <sup>-</sup> emission	Photoelectric e <sup>-</sup> emission
Energy per inelastic event	20 eV	8 keV
Energy per elastic event	60 eV	80 eV
Energy relative to electrons		
inelastic (Compton)	1	400
elastic (Rayleigh)	1	1000

# Negative Stain and Cryo

- Negative stain (usually 0.5% uranyl acetate)
  - Easy to prepare
  - Good contrast
  - Preservation
  - Sample distortion
  - Resolution limited to about 20 angstroms
- Cryo
  - Difficult sample prep
  - Low contrast
  - Best preservation and therefore resolution

# Negative staining

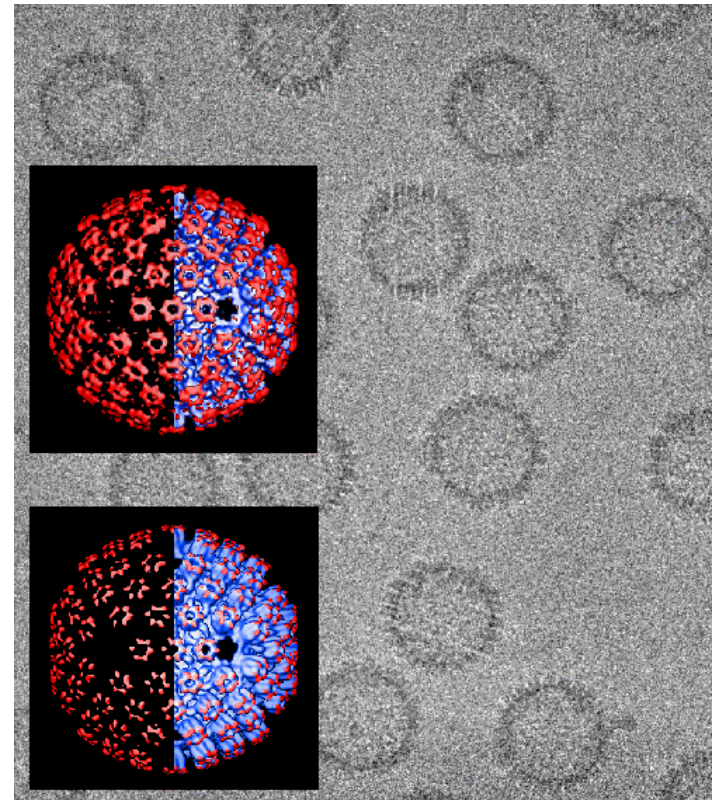
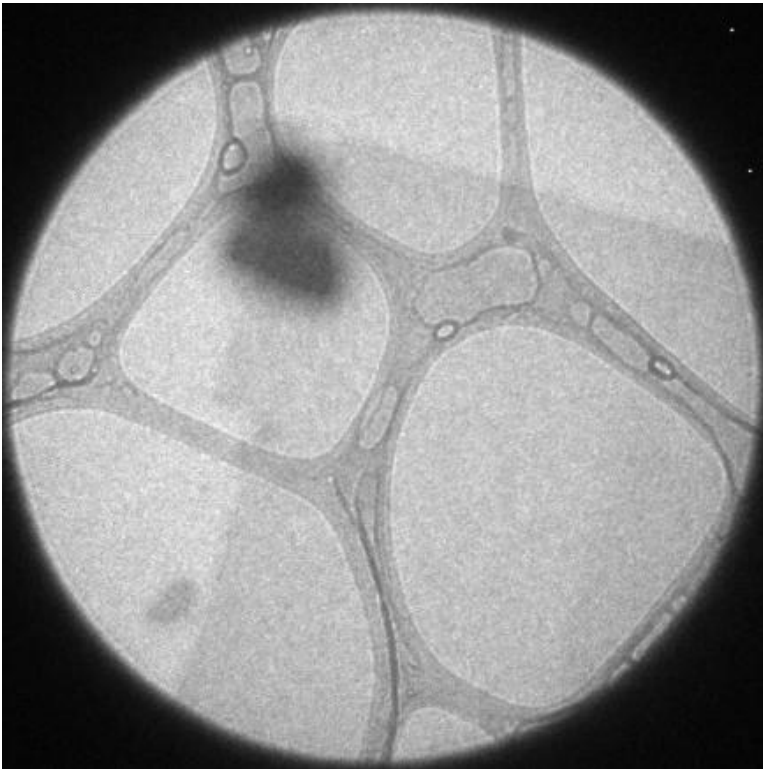


Bob Horne (Cambridge)

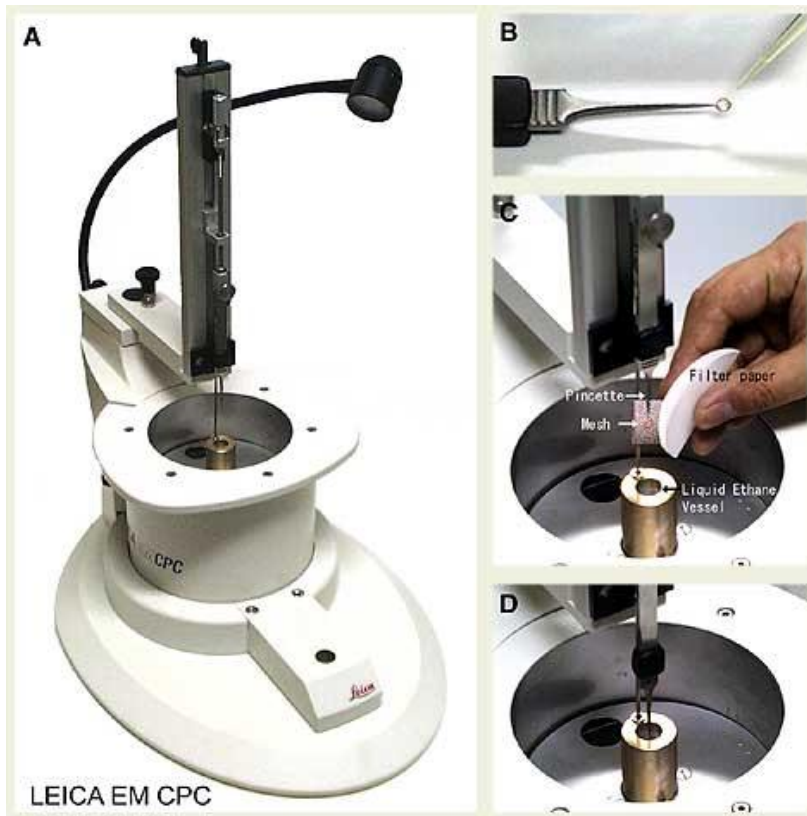
# Cryo prep using holey film

H. Fernandez-moran  
B. Glaeser  
K. Taylor  
J. Dubochet

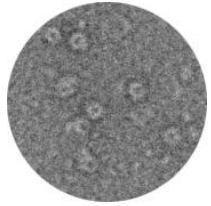
Aaron Klug



# Flash freeze in liquid ethane



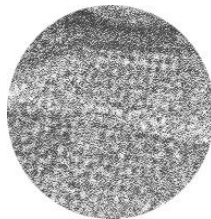
# Samples



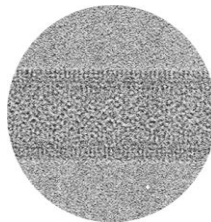
- Single Particles (Proteins, Ribosome)
  - No crystallization
  - Weak amplitude, no diffraction, alignment ambiguity, particle flexibility
  - ~7 angstroms



- Fibers and filaments (tubulin, collagen)
  - No crystallization, 2D distortion corrections, phase restrictions
  - Weak amplitude, no diffraction
  - ~9 angstroms



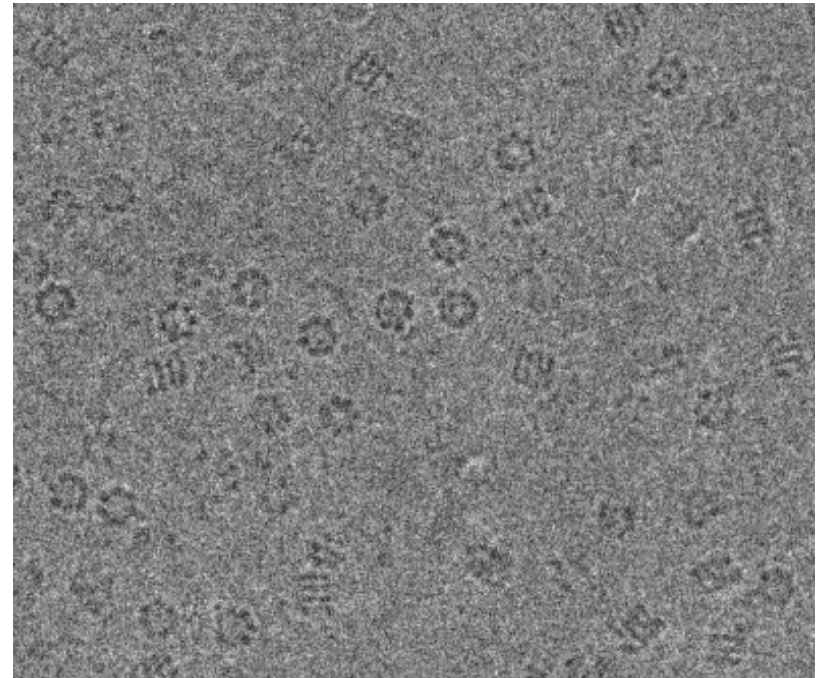
- 2D crystals (BR, AQP, LHCII)
  - Diffraction amplitudes, 2D distortion corrections, crystallographic methods
  - Crystallization, many tilts required, anisotropic data
  - ~3 angstroms



- Tubular crystals (AChR, Ca<sup>++</sup>-ATPase)
  - Crystallization, No diffraction
  - Isotropic data, 3D distortion corrections, phase restrictions
  - ~5 angstroms

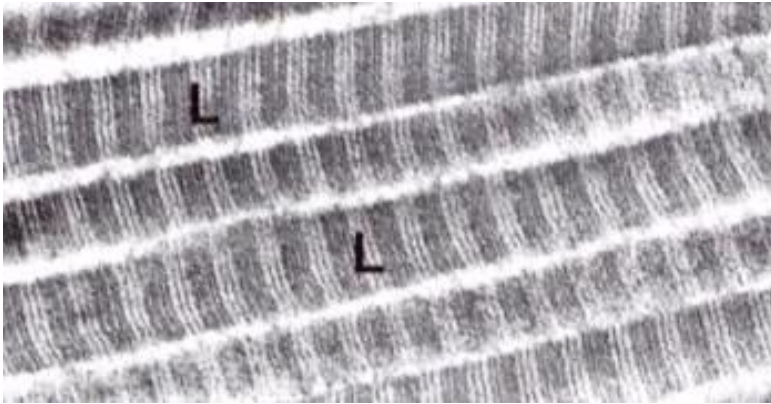
# Single particles

- Applicable to any protein or protein complex  $> 50\text{kD}$
- Most common sample
- Number of software suites available
- Resolution  $\sim 9\text{\AA}$  ( $< 7$  with symmetry)





# Fibers and filaments

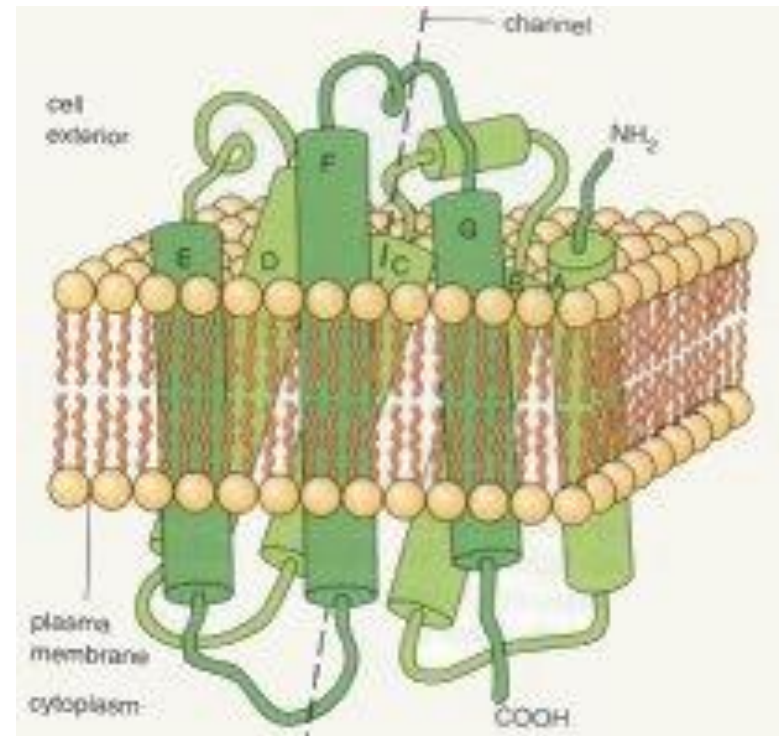
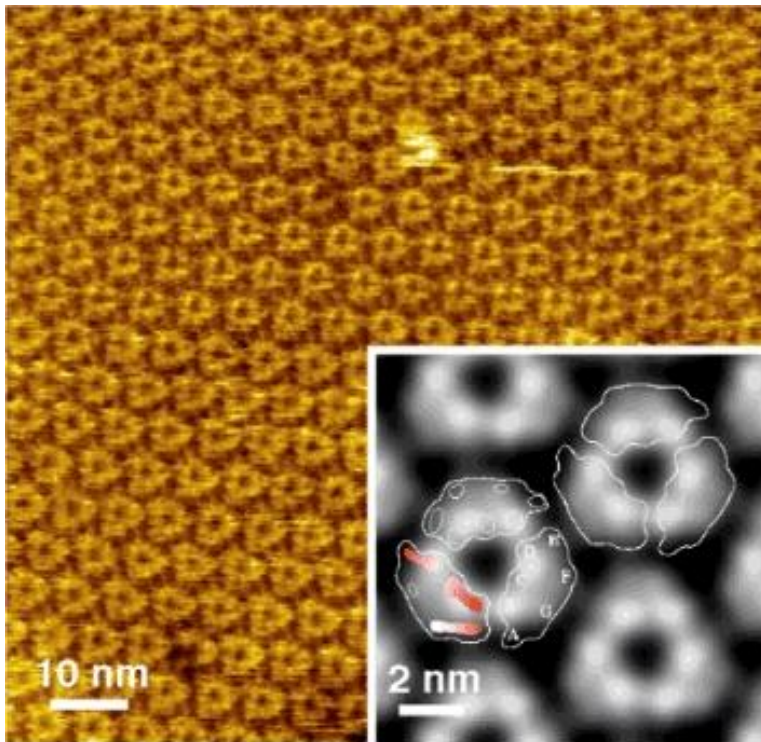


DNA, collagen, etc



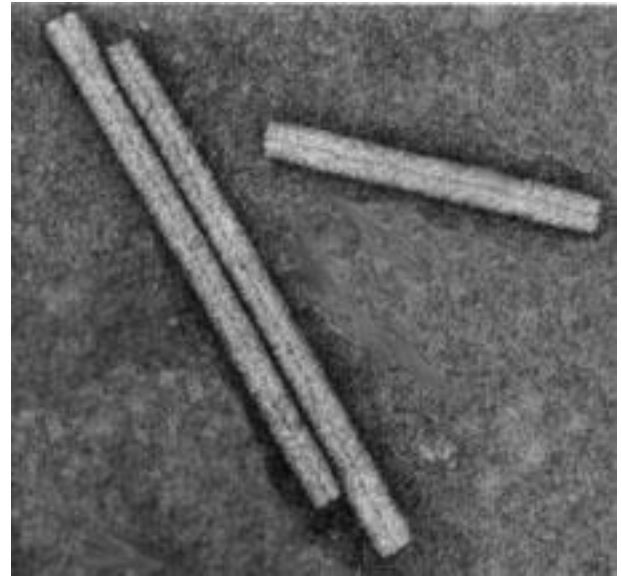
# 2D Xtals

Henderson and Unwin

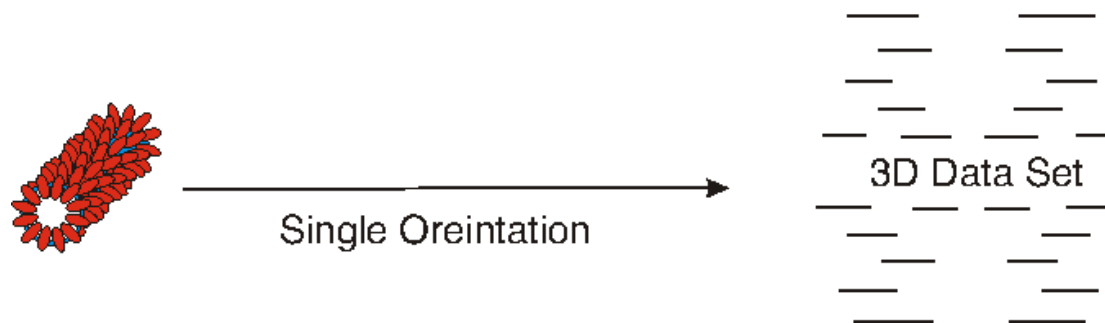
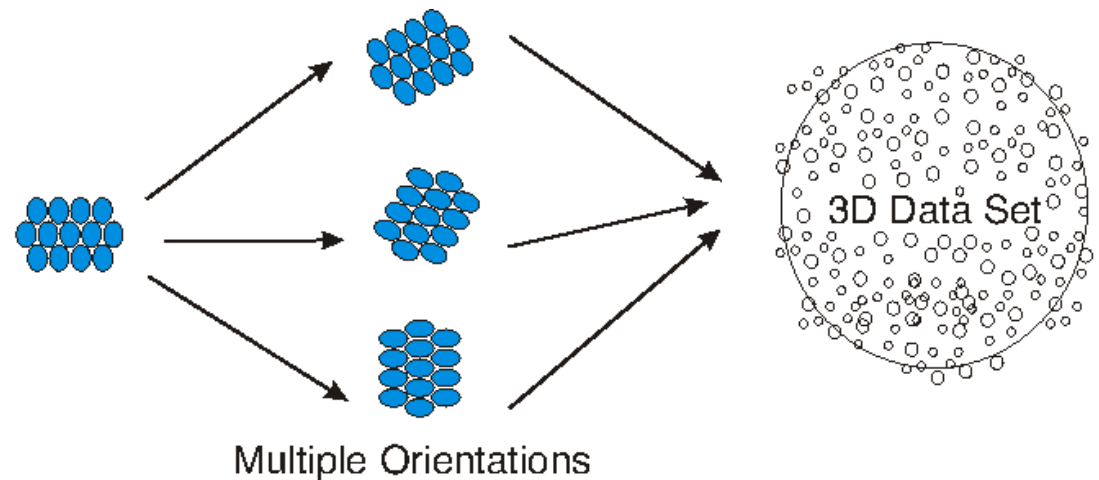


# Tubular crystals

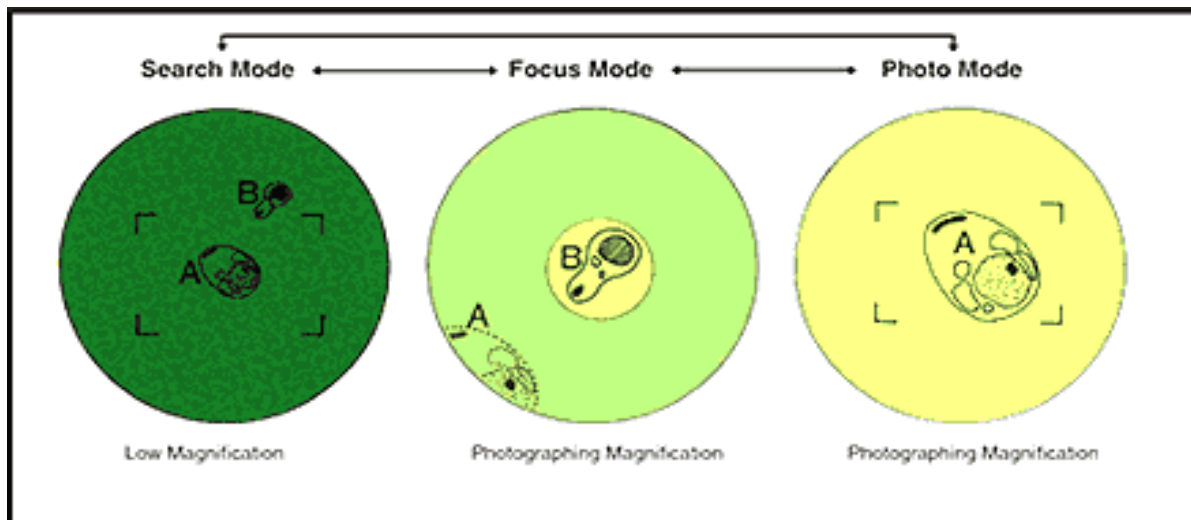
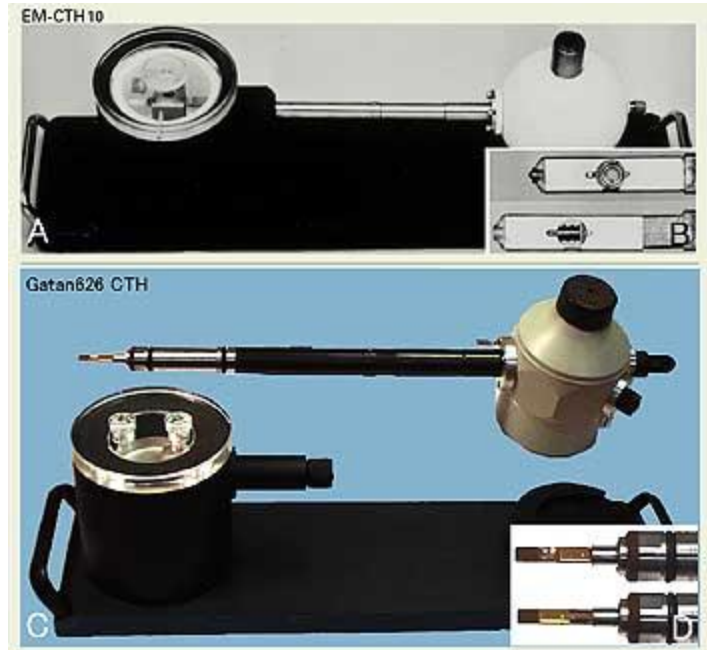
**Rolled 2D xtal**



# Tubular xtal versus 2D or 3D xtal



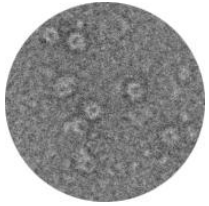
# Data collection



# Image recording

- Film
  - High density content (~20kx16k pixels)
  - Slow (development time, drying)
  - Requires digitization (scanning takes hours)
- CCD
  - Low density content (4kx4k pixels)
  - Fast (ms to sec)
  - Direct digital

# Processing data



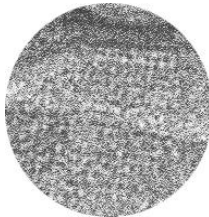
- Single Particles (Proteins, Ribosome)

- Pick particles
- Align
- Classify, average and reconstruction



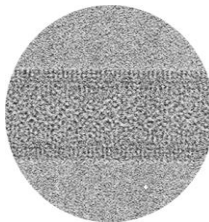
- Fibers and filaments (tubulin, collagen)

- Pick segments determine symmetry
- Align/rotate
- Average



- 2D crystals (BR, AQP, LHCII)

- Process images to achieve phases
- Process diffraction data for amplitudes
- Combine and refine as in X-ray



- Tubular crystals (AchR, Ca<sup>++</sup>-ATPase)

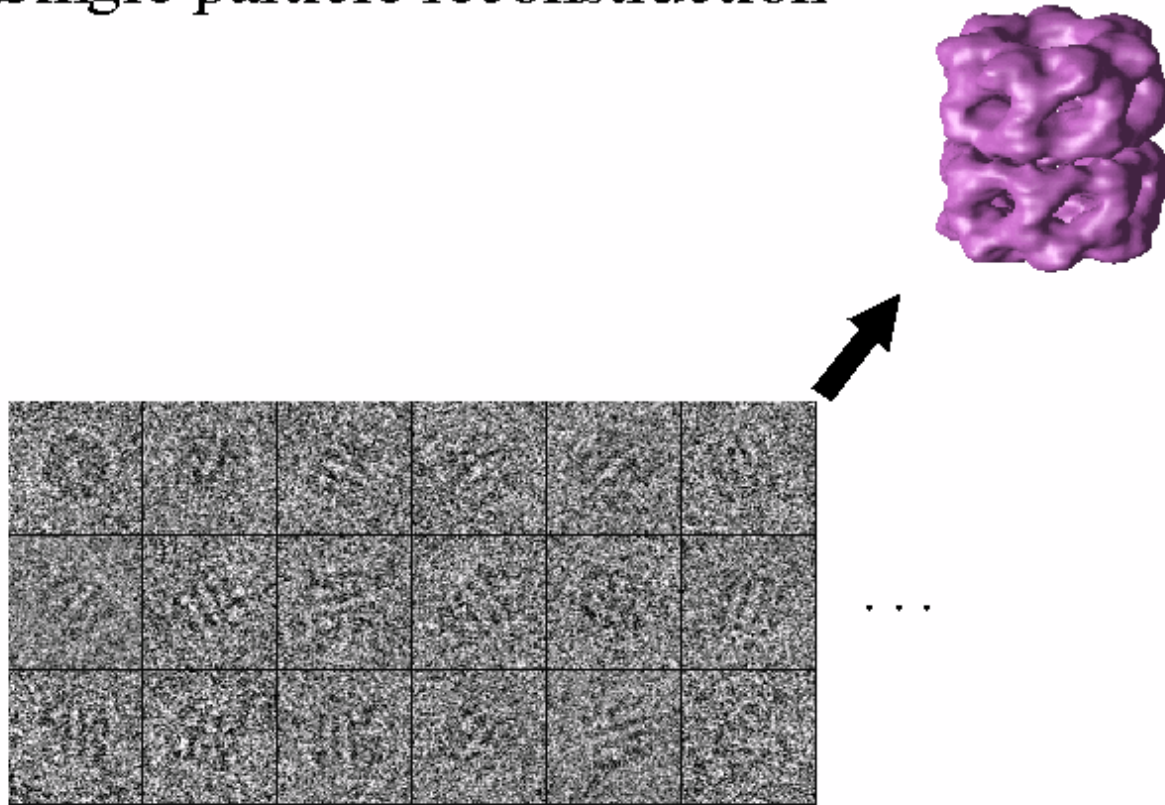
- Determine tube symmetry
- Pick segments and distortion correction
- Average and sum segments



# Data processing 1: single particle

Mostly swiped from Steve Ludtke's web site <http://ncmi.bcm.tmc.edu/~stevel/EMAN/doc/>

## Single particle reconstruction

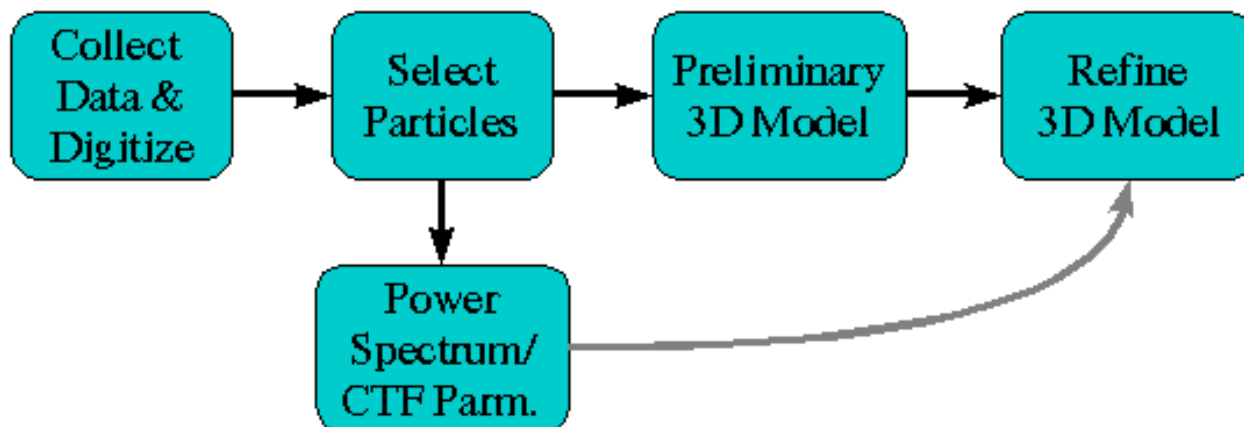




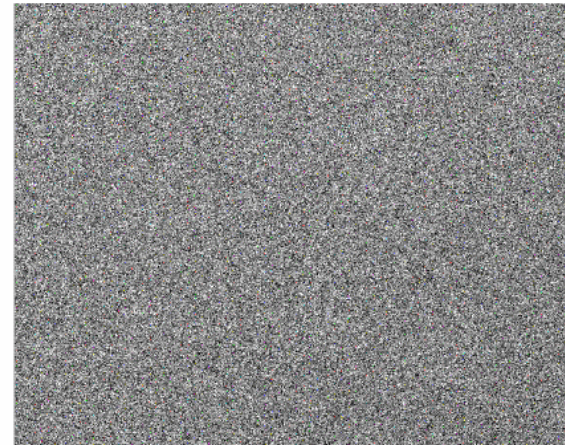
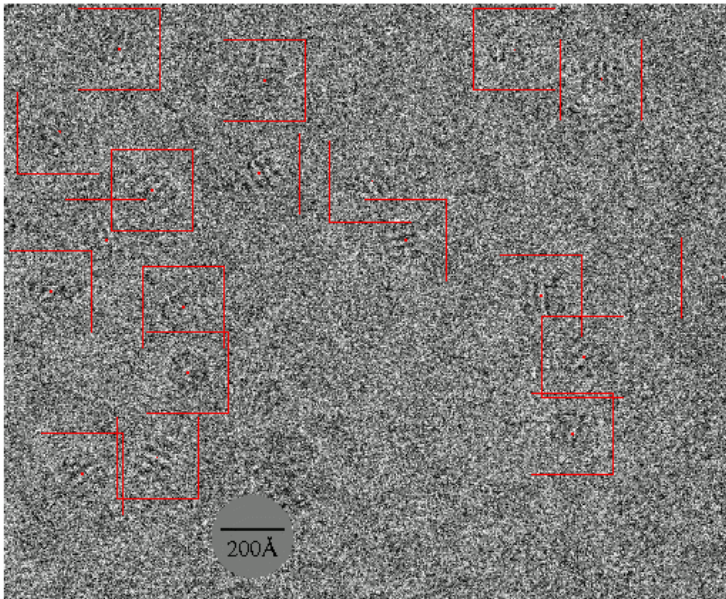
# Software

- Spider
  - [http://www.wadsworth.org/spider\\_doc/spider/docs/](http://www.wadsworth.org/spider_doc/spider/docs/)
  - \$1500 + \$300/yr for updates, with source
- Imagic
  - <http://www.imagescience.de/imagick/welcome.htm>
  - Commercial package, ~ \$6000/yr
- EMAN
  - <http://ncmi.bcm.tmc.edu/~steve/EMAN/doc>
  - Free, complete with C++ source

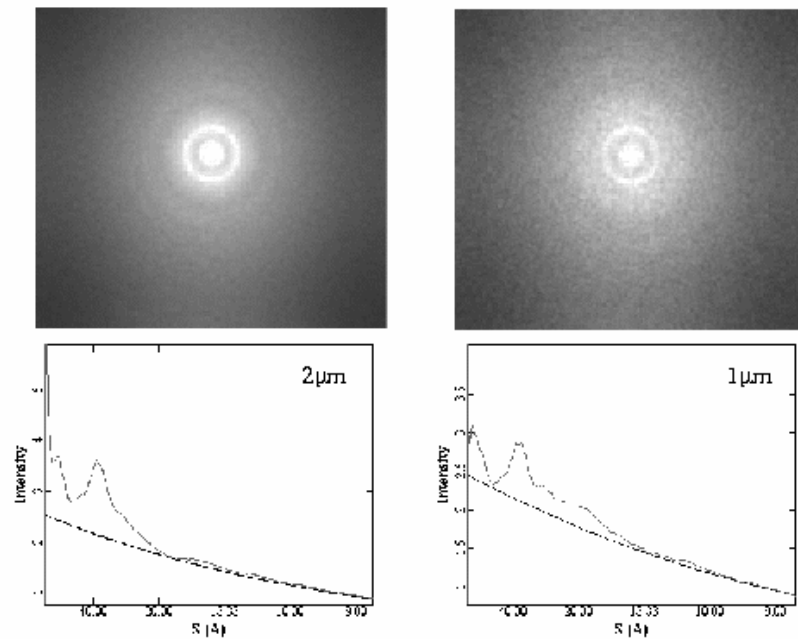
# The Reconstruction Process



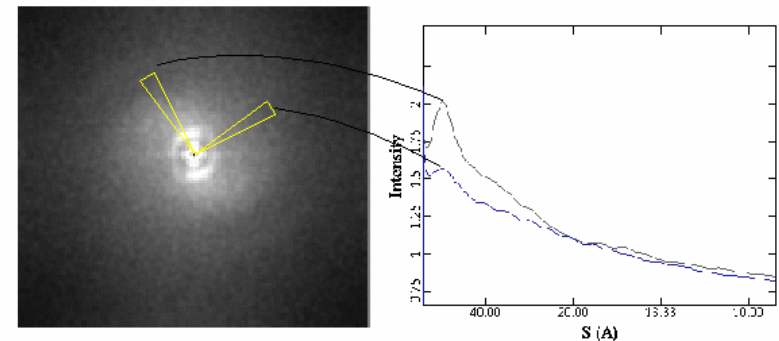
# Pick particles (manual or semiauto)



## Evaluate Particles



## Astigmatism &/or Drift



**Looking for astigmatism, drift, charging etc....**

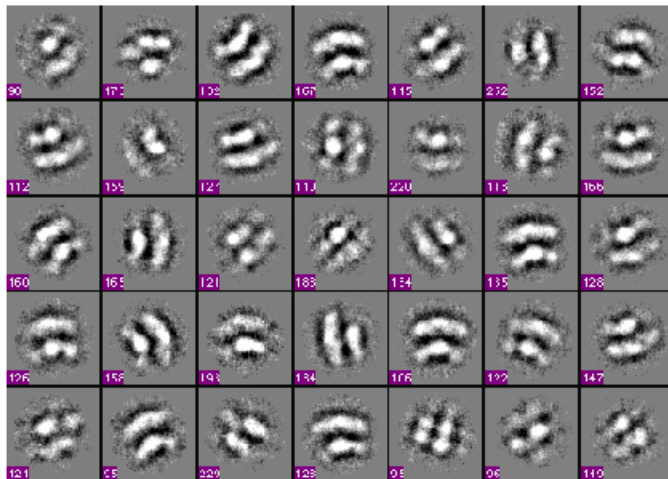
# Now on to the first model

- First rule of thumb...be cautious...
- How to classify particles
  - Reference free classification and alignment
  - MSA
- Application of symmetry
- Random conical tilt

# Reference free classification MSA

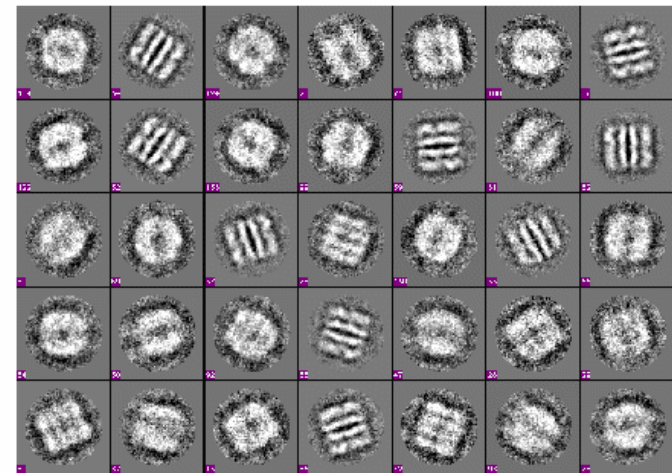
FAS

reference free class averages



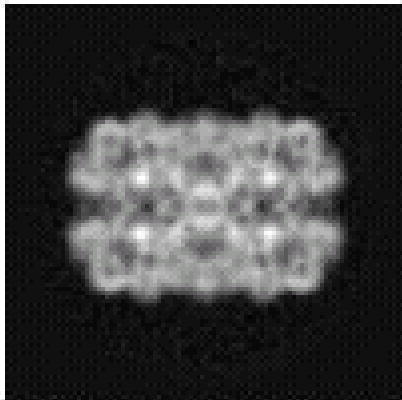
GroEL

reference free class averages

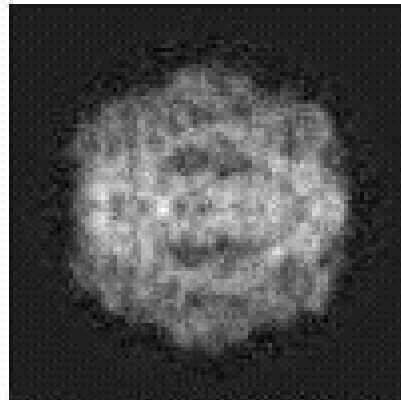


**Can we tell the symmetry a priori???**

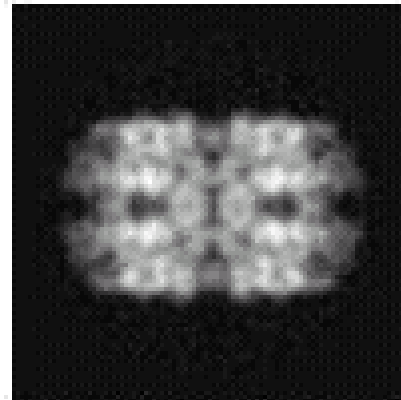
MSA.....variance....(SD)<sup>2</sup>



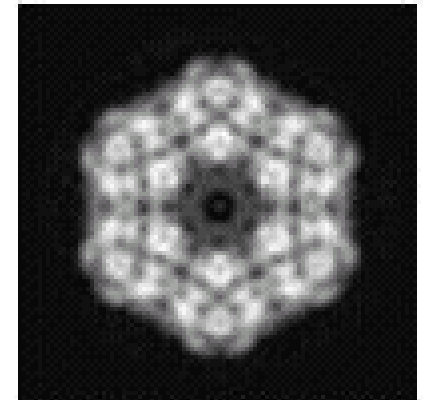
CLAVG001



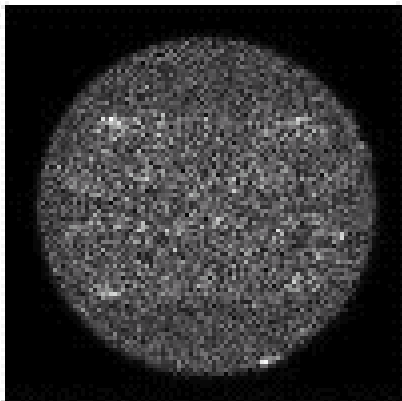
CLAVG002



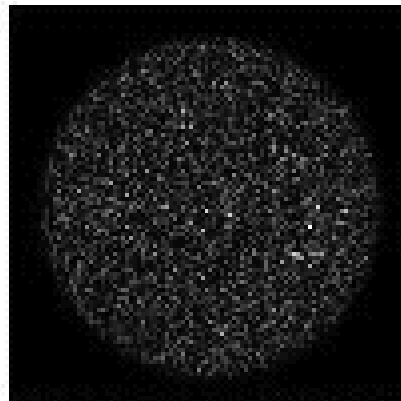
CLAVG003



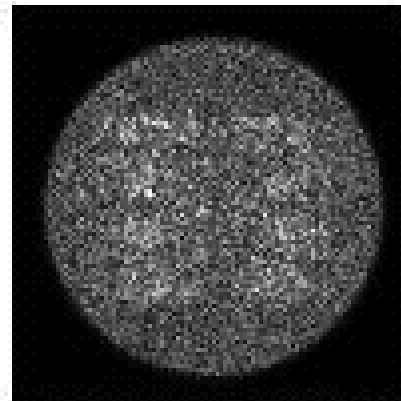
CLAVG004



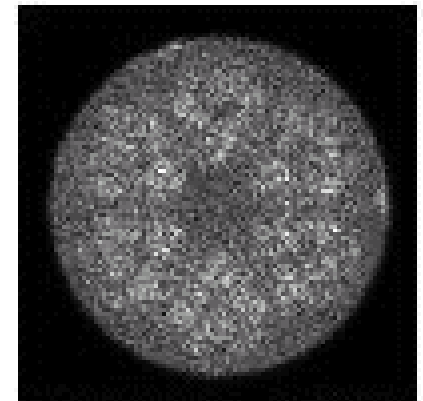
CLVAR001



CLVAR002



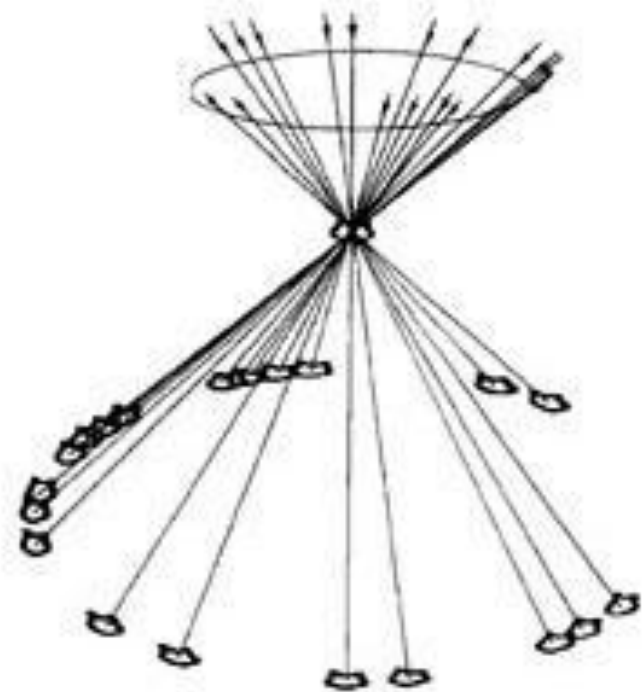
CLVAR003



CLVAR004

# Random conical tilt

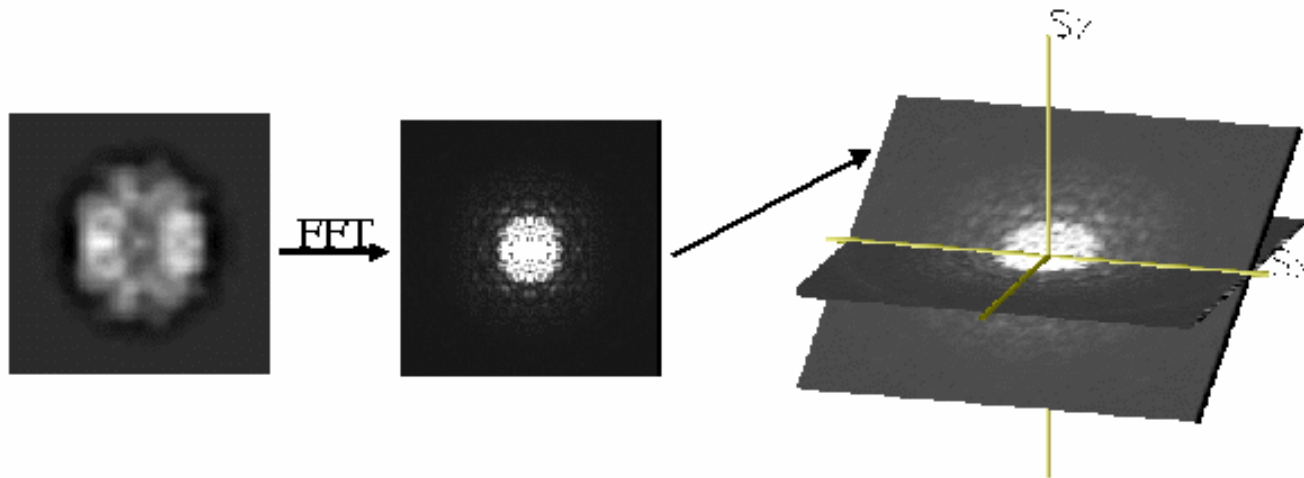
- Image pairs taken of the same sample with an angular tilt applied between them
- Determine particle pairs and construct reference model





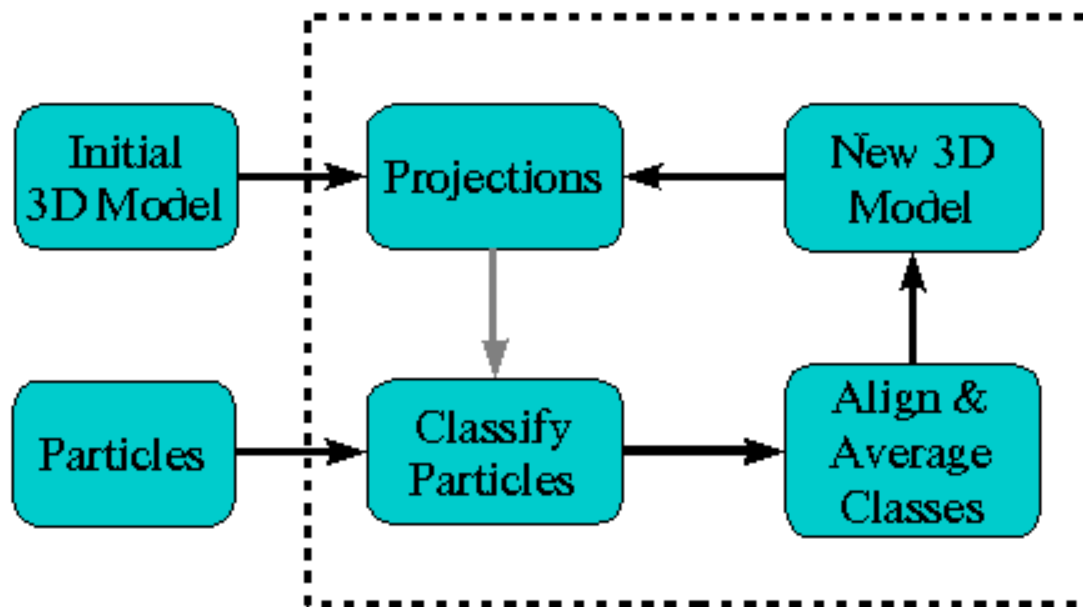
# Use of common lines to align different orientations

## Reconstruction & the Asymmetric Triangle

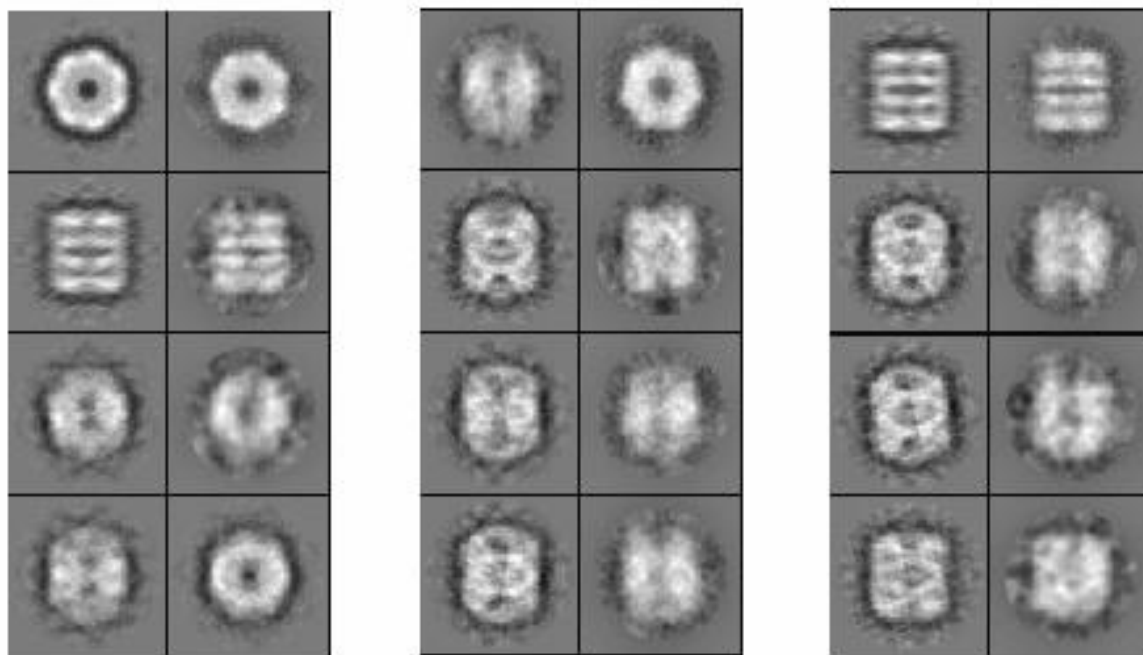


# Refinement

## Refinement - EMAN

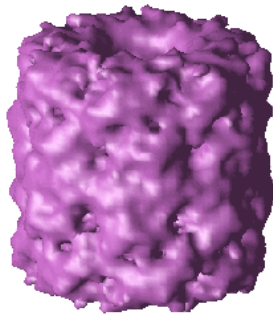


# Class Averages

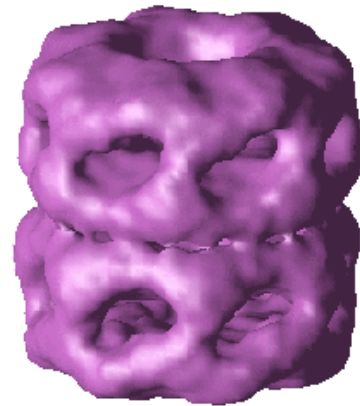


# Multiple rounds of refinement...

Iteration 1

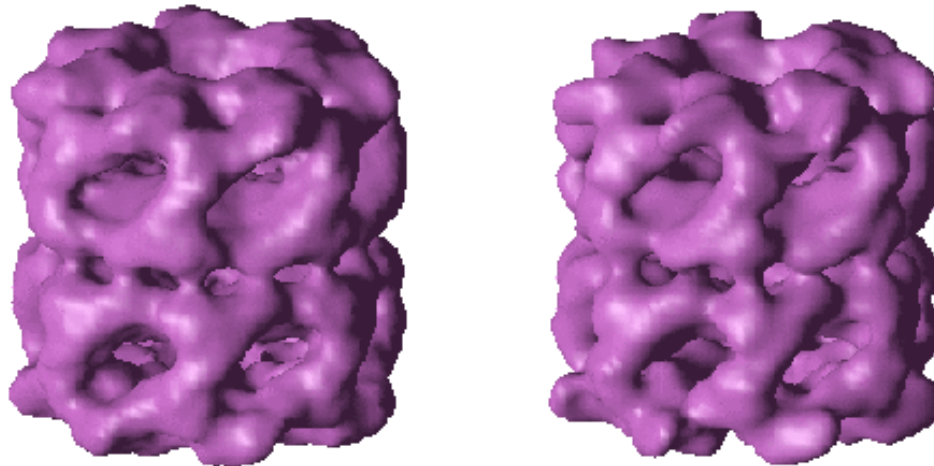


Iteration 3



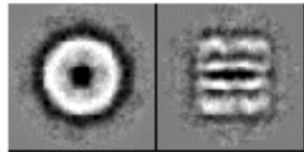
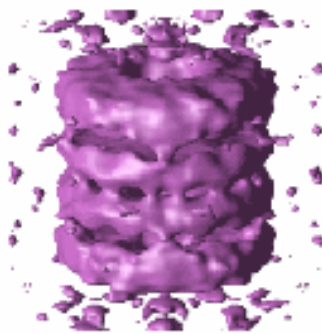
# Convergence when no improvement in the alignment statistics

Final vs. x- ray

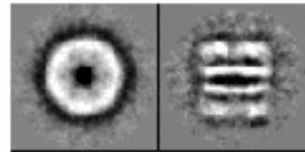
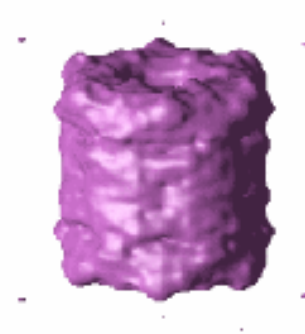


# Try different symmetries

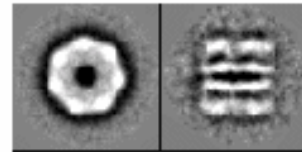
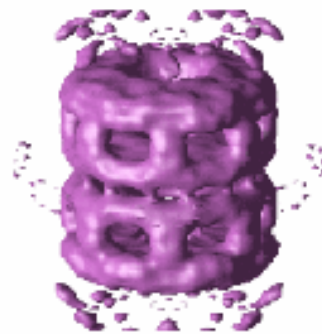
Check Other Possibilities



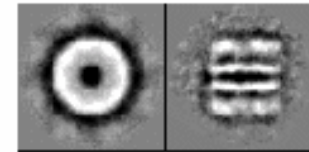
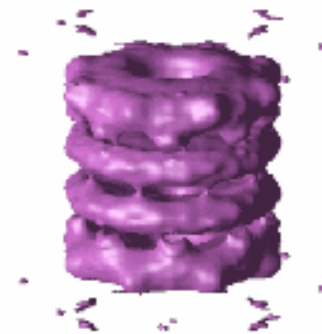
D5



D6

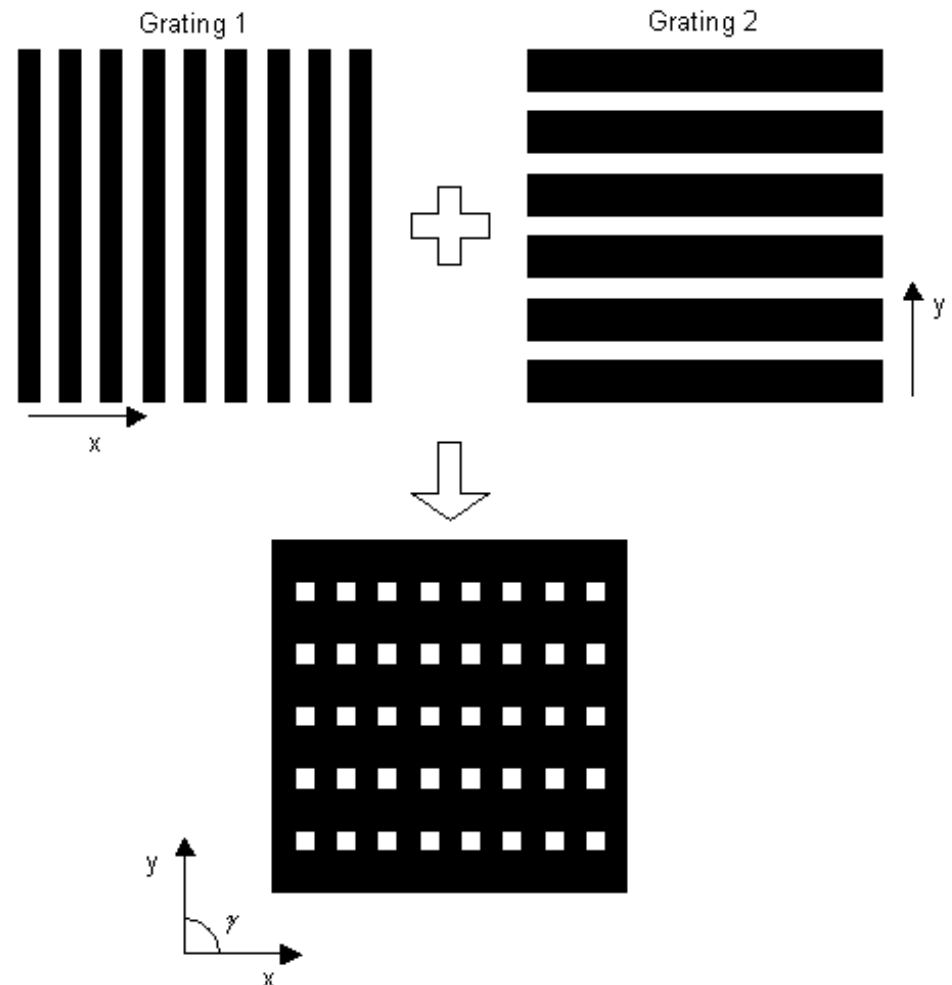
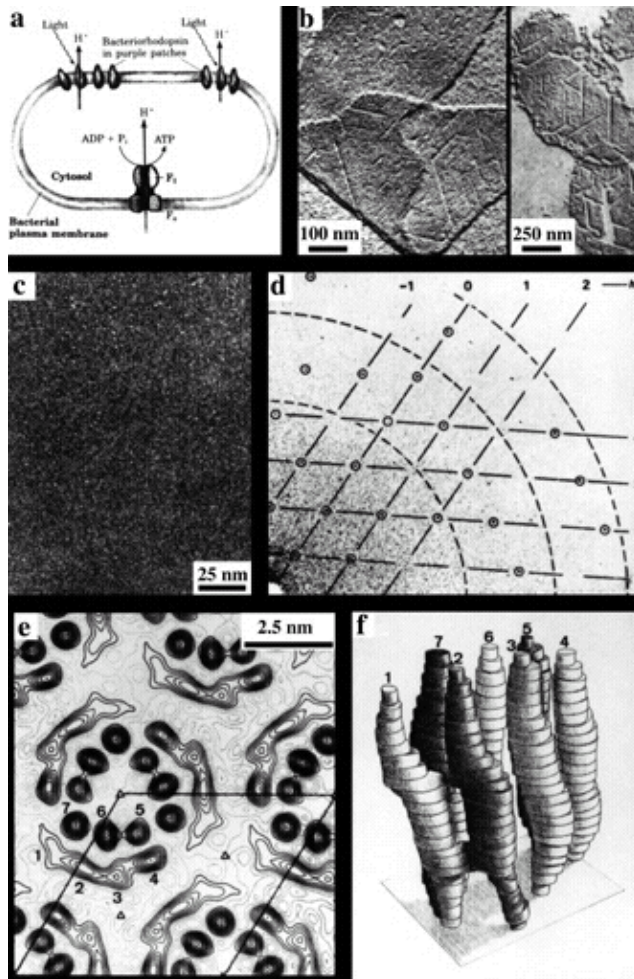


D7



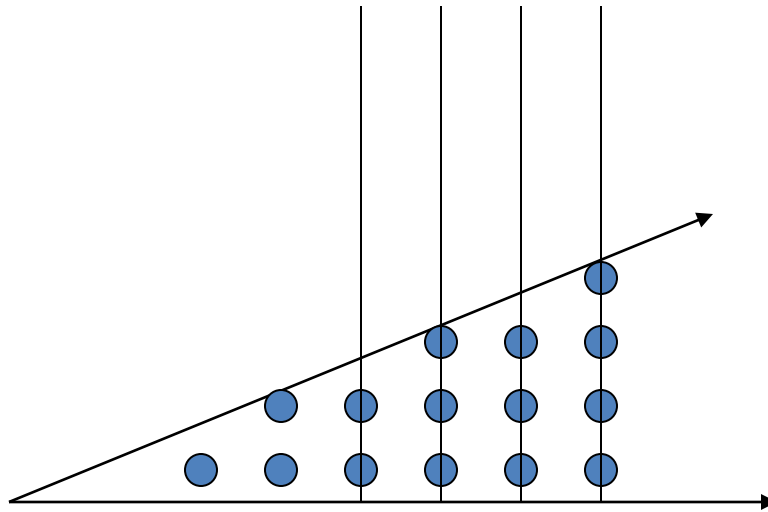
D8

# Data processing 2: 2D xtal



# Why lattice lines?

**Z dimension has an effective real space  $D$  of infinity  
Hence in reciprocal space the lattice spacing is 0**





**Table 1.** Electron crystallographic table*A. Crystallographic parameters*

Space group (layer group)

*P*3 (*p*3)

Lattice constants

 $a = b = 62.45 \text{ \AA}$ ,  $\gamma = 120^\circ$  $c = 100 \text{ \AA}$  (assumed in refinement)

Thickness

70  $\text{\AA}$  (used in LATLINE)*B. Electron diffraction (amplitude information)*

No. of diffraction patterns

339

Resolution limit used ( $\text{\AA}$ )

3.0

Maximum tilt angle (deg.)

70.6

No. of observed reflections

110,812

Friedel *R*-factor (%)

17.6

*C. Electron microscopy (phase information)*

No. of images

181

Resolution limit used ( $\text{\AA}$ )

3.0

Maximum tilt angle (deg.)

61.2

No. of observed reflections

25,225

*D. Merged data*Resolution ( $\text{\AA}$ )

3.0

No. of unique reflections

6892 (with amplitudes and phases)

Merging *R*-factor (%)

31.3

Phase residual (deg.)

46.8

Completeness (%)

78.4

*E. Refinement*

No. of degrees of freedom

6672 (without hydrogen atoms)

*R*-factor (%)

23.7

Free *R*-factor (%)

33.0

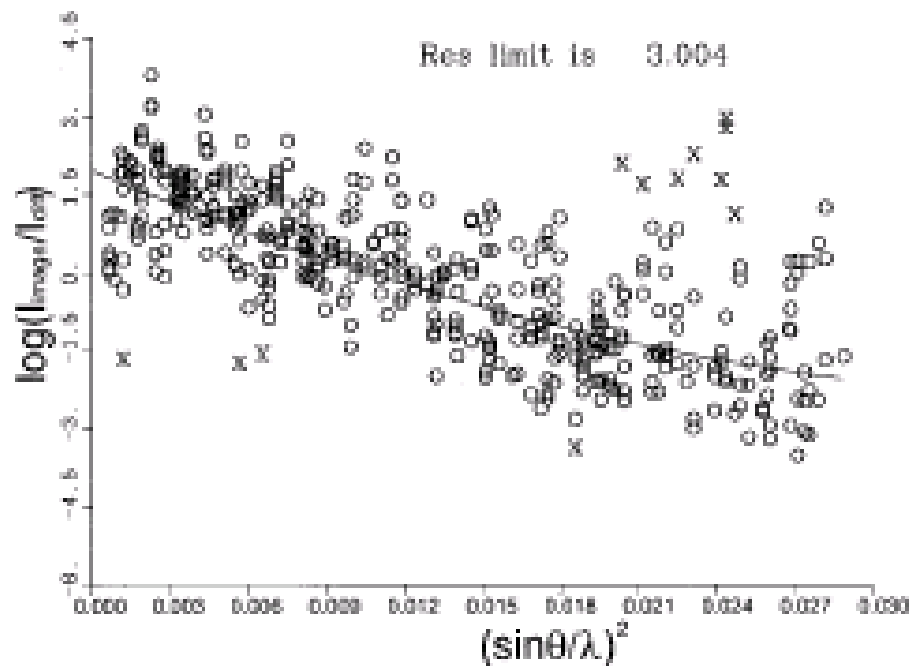
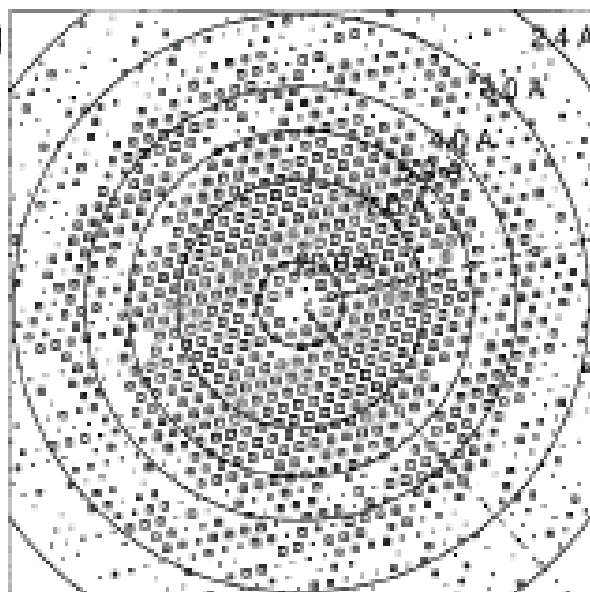
Phase residual (deg.)

54.4

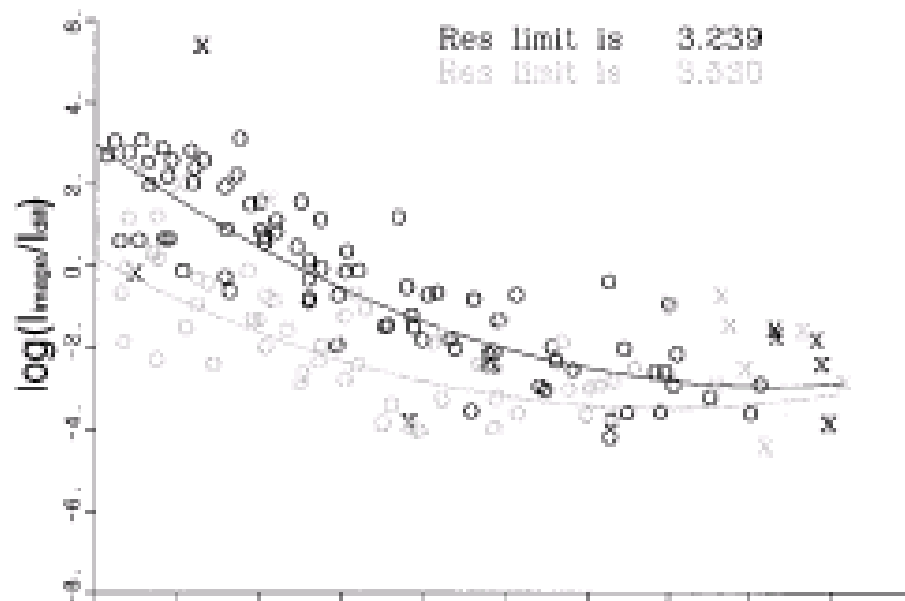
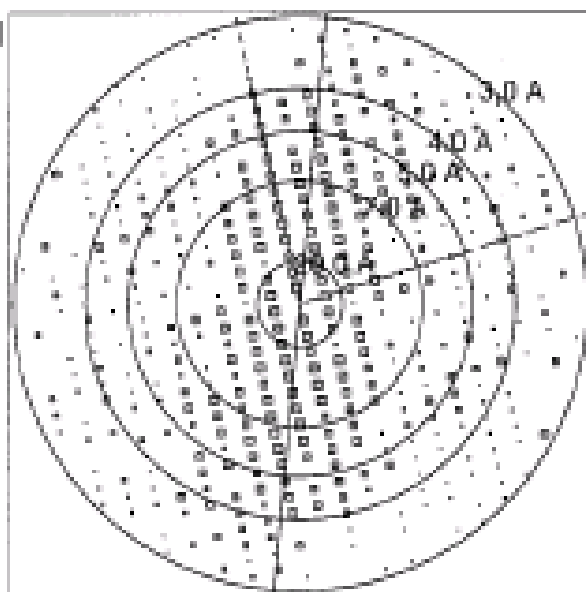
Free phase residual (deg.)

63.3

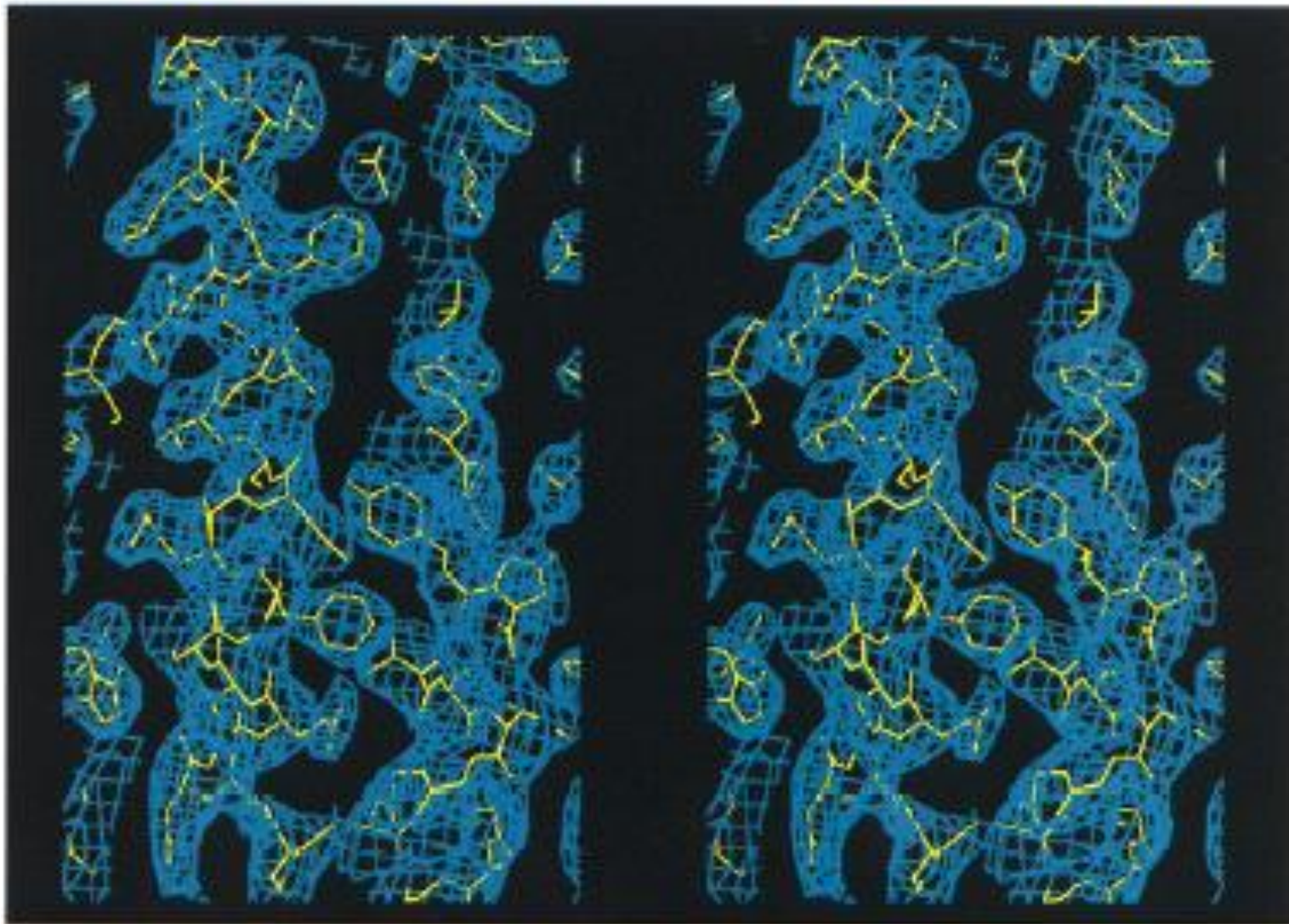
0 deg



60 deg



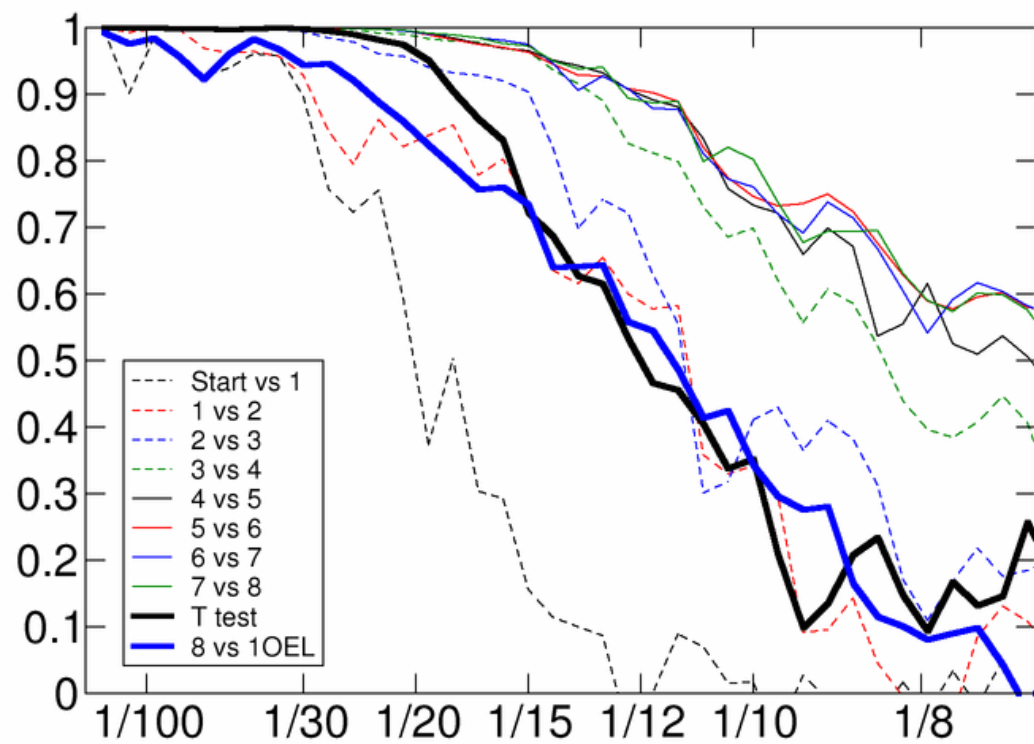
# A well refined EM map



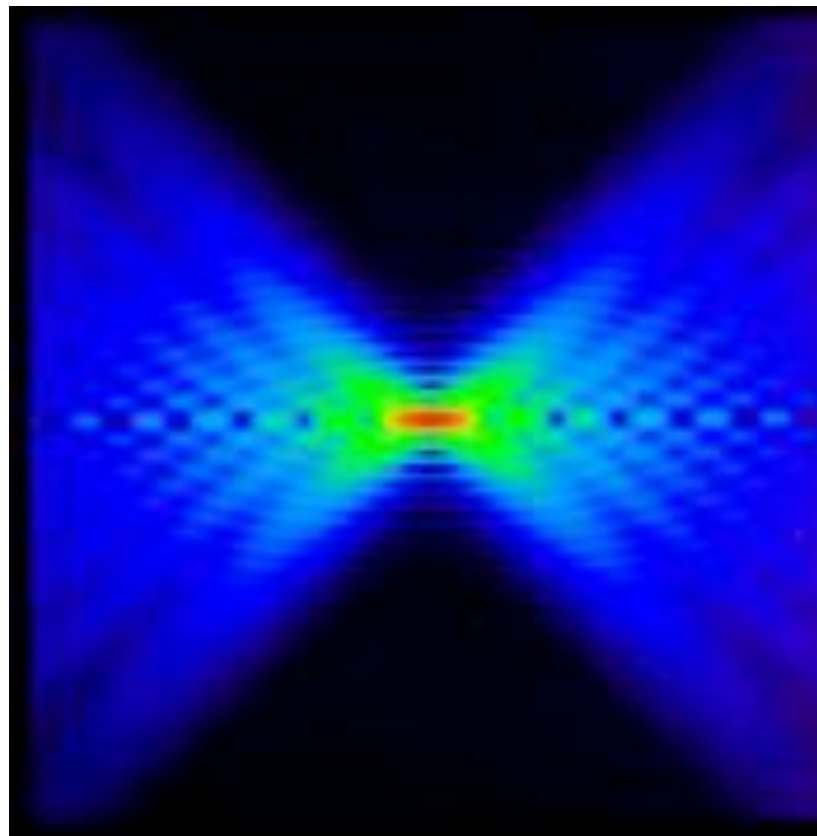
# Resolution and Resolvability

- Single particles, filaments, tubes
  - FSC
    - Which criteria to use (0.5 or 3 sigma)
- 2D xtals diffraction (like X-ray)
  - But anisotropy or point spread function

# FSC



# Point spread function



# Resolution vs Resolvability

- Resolution is a calculated value
  - FSC or measured amplitudes above a certain sigma value.
- Resolvability is a perceived value
  - What can I see in the map
    - Is a 4 angstroms map really 4 angstroms if one cannot discern beta sheet structure?